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TITLE OF THE INVENTION

POSITIVE ELECTRODE ACTIVE MATERIAL AND LITHIUM ION SECONDARY BATTERY

CROSS-REFERENCE TO RELATED APPLICATIONS

This application is based upon and claims the benefit of priority from prior Japanese Patent Application No. 2001-014891, filed January 23, 2001, the entire contents of which are incorporated herein by reference.

BACKGROUND OF THE INVENTION

1. Field of the Invention

The present invention relates to a positive electrode active material containing a lithium-containing nickel oxide and to a lithium ion secondary battery equipped with the positive electrode active material.

2. Description of the Related Art

In recent years, electronic appliances such as portable personal computers and portable telephones are being miniaturized and made lightweight, and miniaturization and reduction of weight are being required for the secondary batteries used as power sources for these electronic appliances.

A lithium ion secondary battery using as a material of the negative electrode a substance capable of absorbing-desorbing lithium ions such as a carbon material has been developed as a secondary battery

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meeting the requirement described above, and has been put to a practical use as a power source for small electronic appliances. The secondary battery is smaller and lighter, and has a higher energy density, than a conventional lead accumulator or nickel-cadmium battery, and the demands for this particular secondary battery is on the increase.

A lithium-containing cobalt oxide (LiCoO₂), which permits a high discharge potential and a high energy density to be obtained, has been put to a practical use as a positive electrode active material of the lithium ion secondary battery. However, the amount of natural resources of cobalt used as the raw material of the complex compound is very small, and the ore deposit that can be commercially utilized is unevenly distributed in a small number of countries. As a result, cobalt is costly and invites a large fluctuation in price. It follows that the cobalt supply in the future will be unstable.

Under the circumstances, researches into positive electrode active materials other than lithium-containing cobalt oxide have been vigorously carried out in recent years. For example, various compounds are reported in respect of the composite oxides between lithium and manganese, which are synthesized by various manganese raw materials and lithium raw material. To be more specific, lithium manganese composite oxide

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represented by LiMn₂O₄ having a Spinel type crystal structure is allowed to exhibit 3 V of potential relative to lithium by electrochemical oxidation and has a theoretical charge-discharge capacity of 148 mAh/q.

However, the lithium ion secondary battery using a manganese oxide or a lithium manganese composite oxide as the positive electrode active material gives rise to the defect that, when the secondary battery is used under an environment not lower than room temperature, the deterioration in the capacity of the secondary battery is markedly increased. It should be noted that the manganese oxide or the lithium manganese composite oxide is rendered unstable under high temperatures so as to cause Mn to elute into the nonaqueous electrolyte, giving rise to the defect noted above. Particularly, a large lithium ion secondary battery has been developed in recent years in various technical fields for use in an electric automobile or in road leveling. In the large lithium ion secondary battery, the heat generation during the use of the secondary battery is rendered non-negligible with enlargement in the size of the lithium ion secondary battery, with the result that the temperature inside the battery tends to be rendered relatively high even if the ambient temperature is close to room temperature. Also, even when it come to a relatively

small battery used in, for example, a small portable electronic appliance, it is possible for the battery to be used under a high temperature environment such as within a room of a vehicle in the midsummer, with the result that it is possible for the temperature inside the battery to be rendered relatively high. Under the circumstances, it is very difficult to put a positive electrode active material using manganese as a raw material to practical use.

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Researches on the nickel composite oxides are being vigorously carried out as post cobalt composite oxides. A nickel composite oxide, e.g., LiNiO2, exhibits a theoretical capacity of 180 to 200 mAh/q, which is larger than that of each of the LiCoO2 series active material and the LiMn₂O₄ series active material. In addition, LiNiO2 has an optimum discharge potential of about 3.6 V on the average and, thus, provides a highly hopeful positive electrode active material. However, the crystal structure of LiNiO2 is unstable, giving rise to the problem that the initial discharge capacity is greatly lowered in the charge-discharge cycle test with increase in the number of the chargedischarge cycles, and to the additional problem in safety that rupture and ignition are brought about in the nail sticking test of the lithium ion secondary battery prepared by using LiNiO2.

On the other hand, claim 1 of Japanese Patent

Disclosure (Kokai) No. 10-326621 recites a secondary battery comprising a positive electrode using a lithium-containing metal composite oxide as an active material, a negative electrode using a metal composite oxide having an amorphous structure, and a nonaqueous electrolyte. Used as the active material of the positive electrode is a nickel-containing lithium composite oxide represented by $\text{Li}_X \text{Ni}_{1-y} \text{M}_y \text{O}_{2-z} \text{Xa}$, where M represents at least one kind of an element selected from the group consisting of the elements of Group 2, Group 13, Group 14 of the Periodic Table and the transition metals, X represents a halogen atom, and x, y, z and a are defined to be 0.2 < x \leq 1.2, $0 \leq y \leq 0.5$, $0 \leq z \leq 1$ and $0 \leq a \leq 2z$.

Also, the heading [0010] of the Japanese Patent document quoted above refers to the desirable concentrations of impurities other than the elements Li, Ni, Co and M contained in the positive electrode active material represented by the structural formula given above. Specifically, the Japanese Patent document quoted above teaches that it is desirable for the concentration of the impurity Fe to be not higher than 0.01% by weight (not higher than 100 ppm), for the concentration of the impurity Cu to be not higher than 0.01% by weight (not higher than 100 ppm), and for the concentration of each of the impurities Ca, Na and sulfate group (SO₄) to be not higher than 0.05% by

weight (not higher than 500 ppm). Also, the Japanese Patent document quoted above teaches that it is desirable for the concentration of the water (H_2O) to be not higher than 0.1% by weight.

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What should be noted is that the elements other than those specified in the structural formula, i.e., Fe, Cu, Na and sulfate group SO₄, are impurities in the Japanese Patent document in question and, thus, should desirably be contained in a smaller amount.

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BRIEF SUMMARY OF THE INVENTION

An object of the present invention is to provide a positive electrode active material capable of eliminating rupture and ignition in the nail sticking test and also capable of improving the large discharge characteristics (discharge rate characteristics) and to provide a lithium ion secondary battery equipped with the particular positive electrode active material.

According to a first aspect of the present invention, there is provided a lithium ion secondary battery, comprising:

- a positive electrode comprising an active material containing a composite oxide;
 - a negative electrode; and
 - a nonaqueous electrolyte;

25 the composite oxide having a composition represented by a structural formula (1) given below:

$$\text{Li}_{X}(\text{Ni}_{1-V}\text{Mel}_{V})(\text{O}_{2-Z}\text{X}_{Z}) + \text{A}$$
 ... (1)

where Mel is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Co, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, y, z are $0.02 \le x \le 1.3$, $0.005 \le y \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm.

According to a second aspect of the present invention, there is provided a lithium ion secondary battery, comprising:

a positive electrode comprising an active material containing a composite oxide;

a negative electrode; and

a nonaqueous electrolyte;

20 the composite oxide having a composition represented by a structural formula (2) given below:

 $\text{Li}_{x}(\text{Ni}_{1-y}\text{Mel}_{y})(O_{2-z}X_{z}) + A + bB$... (2)

where Mel is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Co, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group

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consisting of F, Cl, Br and I, the molar ratios x, y, z are $0.02 \le x \le 1.3$, $0.005 \le y \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm, B contains at least one element selected from the group consisting of Si and Fe, and the content b of the element B in the composite oxide falls within a range of between 20 ppm and 500 ppm.

According to a third aspect of the present invention, there is provided a lithium ion secondary battery, comprising:

a positive electrode comprising an active material containing a composite oxide;

a negative electrode; and

a nonaqueous electrolyte;

the composite oxide having a composition represented by a structural formula (3) given below:

20 $\text{Li}_{x}(\text{Ni}_{1-v-s}\text{Co}_{v}\text{Me2}_{s})(\text{O}_{2-z}\text{X}_{z}) + A$... (3)

where Me2 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, v, s and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 0.5$,

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 $0.005 \le s \le 0.5$ and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm.

According to a fourth aspect of the present invention, there is provided a lithium ion secondary battery, comprising:

a positive electrode comprising an active material containing a composite oxide;

a negative electrode; and

a nonaqueous electrolyte;

the composite oxide having a composition represented by a structural formula (4) given below:

 $\text{Li}_{X}(\text{Ni}_{1-\text{V}-\text{S}}\text{Co}_{V}\text{Me2}_{S})(\text{O}_{2-\text{Z}}\text{X}_{Z}) + \text{A} + \text{bB}$... (4)

where Me2 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, v, s and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 0.5$, $0.005 \le s \le 0.5$ and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm, B contains at

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least one element selected from the group consisting of Si and Fe, and the content b of the element B in the composite oxide falls within a range of between 20 ppm and 500 ppm.

According to a fifth aspect of the present invention, there is provided a positive electrode active material containing a composite oxide having a composition represented by a structural formula (1) given below:

 $\text{Li}_{x}(\text{Ni}_{1-y}\text{Mel}_{y})(O_{2-z}X_{z}) + A$... (1)

where Me1 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Co, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, y, z are $0.02 \le x \le 1.3$, $0.005 \le y \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm.

According to a sixth aspect of the present invention, there is provided a positive electrode active material containing a composite oxide having a composition represented by a structural formula (2) given below:

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 $\text{Li}_{X}(\text{Ni}_{1-V}\text{Mel}_{V})(\text{O}_{2-Z}\text{X}_{Z}) + \text{A} + \text{bB}$... (2)

where Mel is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Co, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, y, z are $0.02 \le x \le 1.3$, $0.005 \le y \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm, B contains at least one element selected from the group consisting of Si and Fe, and the content b of the element B in the composite oxide falls within a range of between 20 ppm and 500 ppm.

According to a seventh aspect of the present invention, there is provided a positive electrode active material containing a composite oxide having a composition represented by a structural formula (3) given below:

 $\text{Li}_{x}(\text{Ni}_{1-v-s}\text{Co}_{v}\text{Me2}_{s})(\text{O}_{2-z}\text{X}_{z}) + \text{A}$... (3)

where Me2 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group

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consisting of F, Cl, Br and I, the molar ratios x, v, s and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 0.5$, $0.005 \le s \le 0.5$ and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm.

According to an eighth aspect of the present invention, there is provided a positive electrode active material containing a composite oxide having a composition represented by a structural formula (4) given below:

 $\text{Li}_{X}(\text{Ni}_{1-v-s}\text{Co}_{v}\text{Me2}_{s}) (\text{O}_{2-z}\text{X}_{z}) + \text{A} + \text{bB}$... (4)

where Me2 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, v, s and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 0.5$, $0.005 \le s \le 0.5$ and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm, B contains at least one element selected from the group consisting of Si and Fe, and the content b of the element B in the

composite oxide falls within a range of between 20 ppm and 500 ppm.

BRIEF DESCRIPTION OF THE SEVERAL DRAWINGS

FIG. 1 is an oblique view, partly broken away, showing a cylindrical lithium ion secondary battery as an example of a lithium ion secondary battery according to one embodiment of the present invention;

FIG. 2 is a cross-sectional view showing a thin lithium ion secondary battery as an example of a lithium ion secondary battery according to another embodiment of the present invention;

FIG. 3 is a cross-sectional view showing in a magnified fashion a portion A shown in FIG. 2; and

FIG. 4 schematically shows the crystal structure of a lithium-containing composite oxide, which is contained in the lithium ion secondary battery for Example 1 of the present invention.

DETAILED DESCRIPTION OF THE INVENTION

The lithium ion secondary battery according to one embodiment of the present invention comprises a positive electrode containing an active material, a negative electrode, and a nonaqueous electrolyte. The active material noted above includes at least one kind of the positive electrode active material selected from the group consisting of first to sixth positive electrode active materials described herein later.

The nonaqueous electrolyte used in the present

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invention includes, for example, a liquid nonaqueous electrolyte prepared by, for example, dissolving a solute in a nonaqueous solvent, a polymer gel-like nonaqueous electrolyte in which a nonaqueous solvent and a solute are held by a polymer material, a polymer solid electrolyte containing a solute as a main component, and an inorganic solid nonaqueous electrolyte having a lithium ionic conductivity. Incidentally, what is described in conjunction with the liquid nonaqueous electrolyte described herein later can be used as the nonaqueous solvent and the solute contained in each of the nonaqueous electrolytes.

The polymer material contained in the gel-like nonaqueous electrolyte noted above includes, for example, polyacrylonitrile, polyacrylate, polyvinylidene fluoride (PVdF), polyethylene oxide (PEO), and polymers containing acrylonitrile, acrylate, vinylidene fluoride or ethylene oxide as the monomer unit. The polymer material contained in the polymer solid electrolyte noted above includes, for example, polyacrylonitrile, polyvinylidene fluoride (PVdF), polyethylene oxide and polymers containing acrylonitrile, vinylidene fluoride or ethylene oxide as a monomer unit. On the other hand, the inorganic solid nonaqueous electrolyte noted above includes, for example, a lithium-containing ceramic material. To be more specific, Li₃N, Li₃PO₄-Li₂S-SiS glass, etc., can

be used as the inorganic solid nonaqueous electrolyte.

An example of a lithium ion secondary battery according to the present invention will now be described.

The lithium ion secondary battery of the present invention comprises a positive electrode containing at least one kind of a positive electrode active material selected from the group consisting of first, second, third, fourth, fifth, and sixth positive electrode active materials, a negative electrode, a separator arranged between the positive electrode and the negative electrode, and a liquid nonaqueous electrolyte impregnated in at least the separator.

Each of the positive electrode, the separator, the negative electrode and the liquid nonaqueous electrolyte will now be described in detail.

1) Positive Electrode

The positive electrode includes a current collector and a positive electrode layer supported by the current collector and containing at least one kind of a positive electrode active material selected from the group consisting of first, second, third, fourth, fifth and sixth positive electrode active materials given below:

(First Positive Electrode Active material)

The first positive electrode active material

contains a composite oxide having a composition

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represented by chemical formula (1) given below:

 $\text{Li}_{X}(\text{Ni}_{1-y}\text{Mel}_{y})(\text{O}_{2-z}\text{X}_{z}) + A$... (1)

where Me1 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Co, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, y, z are $0.02 \le x \le 1.3$, $0.005 \le y \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm.

If the molar ratio x of lithium is less than 0.02, the crystal structure of the composite oxide is rendered highly unstable, with the result that the cycle characteristics of the secondary battery are degraded and the safety is lowered. On the other hand, if the molar ratio x of lithium exceeds 1.3, the discharge capacity and the safety of the secondary battery are lowered. It is more desirable for the molar ratio x of lithium to fall within a range of between 0.05 and 1.2.

25 The molar ratio y of the element Mel should fall within a range of between 0.005 and 0.5. If the molar ratio y is lower than 0.005, it is difficult to improve

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the safety of the secondary battery. On the other hand, if the molar ratio y exceeds 0.5, the discharge capacity of the secondary battery is markedly decreased. It is more desirable for the molar ratio y of the element Mel to fall within a range of between 0.01 and 0.35. Among the elements represented by Mel, it is desirable to use Al, Ti, Mn, Nb or Ta.

The molar ratio z of the halogen element X should fall within a range of between 0.01 and 0.5. If the molar ratio z is less than 0.01, it is difficult to improve the cycle characteristics and the safety of the secondary battery. On the other hand, if the molar ratio z exceeds 0.5, the discharge capacity of the secondary battery is markedly decreased., It is more desirable for the molar ratio z of the halogen element X to fall within a range of between 0.02 and 0.3. Also, among the halogen elements X, it is desirable to use F.

In one of the desired compositions of the composite oxide used in the present invention, the halogen element X contains F, the molar ratio x falls within a range of between 0.05 and 1.2, the molar ratio y falls within a range of between 0.01 and 0.35, and the molar ratio z falls within a range of between 0.02 and 0.3.

The composite oxide used in the present invention contains an element A. The composite oxide containing

the element A is capable of suppressing a rapid increase in the battery temperature when a large current flows through the secondary battery by, for example, a short circuit so as to improve the safety of the secondary battery.

Where Na is contained in the composite oxide, the Na content should fall within a range of between 600 ppm and 3,000 ppm. If the Na content is lower than 600 ppm, it is impossible to obtain a high discharge capacity when the secondary battery is discharged at a high rate. On the other hand, if the Na content exceeds 3,000 ppm, the charge-discharge cycle characteristics of the secondary battery are degraded. It is more desirable for the Na content to fall within a range of between 1,000 ppm and 2,500 ppm. Where the Na content falls within a range of between 1,000 ppm and 2,500 ppm, it is possible to suppress the reduction in the capacity when the charge-discharge cycle in which the discharge is performed at a high rate is repeated.

Where K is contained in the composite oxide, the K content should fall within a range of between 600 ppm and 3,000 ppm. If the K content of the composite oxide is lower than 600 ppm, it is impossible to obtain a high discharge capacity when the secondary battery is discharged at a high rate. On the other hand, if the K content exceeds 3,000 ppm, the charge-discharge cycle

characteristics of the secondary battery are degraded. It is more desirable for the K content to fall within a range of between 1,000 ppm and 2,500 ppm. If the K content is set to fall within a range of between 1,000 ppm and 2,500 ppm, it is possible to suppress the reduction of the capacity when the charge-discharge cycle in which the discharge is performed at a high rate is repeated.

Where S is contained in the composite oxide, the S content should fall within a range of between 600 ppm and 3,000 ppm. If the S content of the composite oxide is lower than 600 ppm, it is impossible to obtain a high discharge capacity when the secondary battery is discharged at a high rate. On the other hand, if the S content exceeds 3,000 ppm, the charge-discharge cycle characteristics of the secondary battery are degraded. It is more desirable for the S content to fall within a range of between 1,000 ppm and 2,500 ppm. If the S content is set to fall within a range of between 1,000 ppm and 2,500 ppm, it is possible to suppress the reduction of the capacity when the charge-discharge cycle in which the discharge is performed at a high rate is repeated.

It is desirable for the element A to include Ca in addition to at least one kind of the element selected from the group consisting of Na, K and S. It is desirable for the Ca content of the composite oxide to

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be not higher than 500 ppm. The Ca content of the composite oxide exceeding 500 ppm tends to promote the deterioration in any of the charge-discharge cycle characteristics, the large current discharge characteristics and the high rate charge-discharge characteristics. It is more desirable for the Ca content to fall within a range of between 20 ppm and 500 ppm. If the Ca content is set to fall within a range of between 20 ppm and 500 ppm, it is possible to improve markedly the large current discharge characteristics (discharge rate characteristics) or the high rate charge-discharge characteristics. It is furthermore desirable for the Ca content to fall within a range of between 50 ppm and 500 ppm.

It is possible to improve sufficiently the safety of the secondary battery even if the elements A are added singly. However, it is possible to further improve the safety of the secondary battery by adding a plurality of kinds of the elements A in combination. The desirable combinations of the elements A include a combination of Na and S, a combination of Ca, Na and S, a combination of Na and Ca, and a combination of S and Ca.

It is desirable for the total content "a" of the elements A in the composite oxide to fall within a range of between 600 ppm and 7000 ppm. If the total content "a" of the elements A is lower than 600 ppm, it

is difficult to improve the large current discharge characteristics of the secondary battery. On the other hand, if the total content "a" of the elements A exceeds 7000 ppm, the charge-discharge cycle characteristics of the secondary battery tends to be degraded. It is more desirable for the total content "a" of the elements A to fall within a range of between 1000 ppm and 5000 ppm.

In the composite oxide represented by the chemical formula (1), it is desirable for at least a part of the element A contains at least one element selected from the group consisting of Na, K, S and Ca to be segregated. Particularly, it is desirable for at least a part of the element A to be precipitated in triple points present in grain boundaries of the composite oxide. If the crystal structure of the composite oxide is constructed to meet the particular requirement, it is possible to further improve both the safety and the cycle characteristics of the secondary battery.

The composite oxide used in the present invention can be manufactured by, for example, a solid state reaction process, a coprecipitation process, or hydrothermal synthesis. Particularly, it is desirable to obtain the composite oxide having a composition represented by chemical formula (1) by firing a powdery mixture of the compounds of each element at 450 to 550°C for 2 to 20 hours with an oxygen gas flow,

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followed by further firing the mixture at 630 to 730° C for 2 to 50 hours with an oxygen gas flow. If the heat treating temperature in the first stage exceeds 550° C or if the heat treating temperature in the second stage exceeds 730° C, the element A melts so as to be attached to the surface of the particle. As a result, the absorption-desorption of lithium is inhibited so as to make it difficult to further improve the safety and the cycle characteristics of the secondary battery. By applying the two stage heat treatment in which the firing temperature and the firing time are defined to fall within the ranges noted above, it is possible to arrange regularly the structure of the oxygen layer-Li layer-oxygen layer-(Ni+Me1) layer-oxygen layer-Li layer and to permit the precipitation of the element A in triple points present in boundaries of the crystal grains. As a result, it is possible to further improve the safety and the charge-discharge cycle

As described above, the lithium ion secondary battery of the present invention comprising a positive electrode containing the first positive electrode active material permits suppressing a rapid elevation of the battery temperature when a large current flows therethrough under the state of a short circuit as in the nail sticking test, making it possible to prevent in advance rupture and ignition. It follows that the

characteristics of the secondary battery.

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safety of the secondary battery can be improved. It is considered reasonable to understand that the element A containing at least one element selected from the group consisting of Na, K, Ca and S serves to lower the series resistance of the positive electrode active material so as to suppress the heat generation caused by the Joule heat derived from the flow of a large current, thereby producing the prominent effect described above. Also, the secondary battery of the present invention makes it possible to realize a large discharge capacity, excellent cycle characteristics and excellent large current discharge characteristics (discharge rate characteristics).

(Second Positive Electrode Active material)

The second positive electrode active material used in the present invention comprises a composite oxide

represented by chemical formula (2) given below:

 $\text{Li}_{X}(\text{Ni}_{1-V}\text{Mel}_{V})(\text{O}_{2-Z}\text{X}_{Z}) + \text{A} + \text{bB}$... (2)

where Me1 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Co, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, y, z are $0.02 \le x \le 1.3$, $0.005 \le y \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and

each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm, B contains at least one element selected from the group consisting of Si and Fe, and the content b of the element B in said composite oxide falls within a range of between 20 ppm and 500 ppm.

The molar ratio x of lithium contained in the composite oxide defined in chemical formula (2) falls within a range of between 0.02 and 1.3, as described above. The reason for the definition of the molar ratio x of lithium is as described previously in conjunction with the composite oxide represented by chemical formula (1) described previously. It is more desirable for the molar ratio x of lithium to fall within a range of between 0.05 and 1.2.

The molar ratio y of the element Mel contained in the composite oxide defined in chemical formula (2) falls within a range of between 0.005 and 0.5, as described above. The reason for the definition of the molar ratio y of the element Mel is as described previously in conjunction with the composite oxide represented by chemical formula (1) described previously. It is more desirable for the molar ratio y of the element Mel to fall within a range of between 0.01 and 0.35. It is desirable to use as the element Mel the elements described previously in conjunction with the first positive electrode active material.

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The molar ratio z of the halogen element X contained in the composite oxide defined in chemical formula (2) falls within a range of between 0.01 and 0.5, as described above. The reason for the definition of the molar ratio z of the halogen element X is as described previously in conjunction with the composite oxide represented by chemical formula (1) described previously. It is more desirable for the molar ratio z of the halogen element X to fall within a range of between 0.02 and 0.3. It is desirable to use F as the halogen element X.

Where Na is contained in the composite oxide defined in chemical formula (2), it is desirable for the Na content to fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Na content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where K is contained in the composite oxide defined in chemical formula (2), it is desirable for the K content to fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for

the K content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

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Where S is contained in the composite oxide defined in chemical formula (2), it is desirable for the S content to fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the S content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

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It is desirable for the element A to include Ca together with at least one element selected from the group consisting of Na, K and S. It is desirable for the Ca content to be not higher than 500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Ca content to fall within a range of between 20 ppm and 500 ppm and, furthermore desirably between 50 ppm and 500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

It is possible to improve sufficiently the safety

of the secondary battery by adding these elements A singly. However, the safety of the secondary battery can be further improved by adding simultaneously a plurality of different kinds of the elements A in combination. The desirable combinations of the elements A are as described previously in conjunction with the first positive electrode active material.

It is desirable for the total content "a" of the elements A in the composite oxide to fall within a range of between 600 ppm and 7000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the total content "a" of the elements A to fall within a range of 1000 ppm and 5000 ppm.

The composite oxide contained in the second positive electrode active material also contains the element B. Where the element B is contained in the composite oxide, it is possible to suppress more effectively the elevation of the battery temperature when a large current flows through the secondary battery by, for example, the occurrence of a short circuit so as to further improve the safety of the secondary battery. However, if the content b of the element B is lower than 20 ppm, it is difficult to improve sufficiently the safety of the secondary battery. On the other hand, if the content b of the

element B exceeds 500 ppm, the charge-discharge cycle characteristics of the secondary battery tend to be markedly lowered. It is more desirable for the content b of the element B to fall within a range of between 20 ppm and 250 ppm. Further, where the element B contains both of Si and Fe, it is possible to further improve the safety and the charge-discharge cycle characteristics of the secondary battery.

In the composite oxide represented by chemical formula (2), it is desirable for at least a part of the element A containing at least one kind of the element selected from the group consisting of Na, K, Ca and S and for at least a part of the element B containing at least one kind of the element selected from the group consisting of Si and Fe to be segregated. Particularly, it is desirable for at least a part of at least one element of the elements A and B to be precipitated in triple points present in grain boundaries of the composite oxide. Where the crystal structure of the composite oxide is constructed in this fashion, it is possible to further improve both the safety and the cycle characteristics of the secondary battery.

The composite oxide used in the present invention can be prepared by, for example, a solid state reaction process, a coprecipitation process or hydrothermal synthesis. Particularly, it is desirable to obtain the

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composite oxide having a composition represented by chemical formula (2) by firing a powdery mixture of the compounds of each element at 450 to 550℃ for 2 to 20 hours with an oxygen gas flow, followed by further firing the mixture at 630 to 730° C for 2 to 50 hours with an oxygen gas flow. If the heat treating temperature in the first stage exceeds $550^{\circ}\mathrm{C}$ or if the heat treating temperature in the second stage exceeds 730 $^{\circ}$ C, the compound of element A and the compound of element B melt so as to be attached to the surface of the particle. As a result, the absorption-desorption of lithium is inhibited so as to make it difficult to improve the safety and the cycle characteristics of the secondary battery. By applying the two stage heat treatment in which the firing temperature and the firing time are defined to fall within the ranges noted above, it is possible to arrange regularly the structure of the oxygen layer-Li layer-oxygen layer-(Ni+Me1) layer-oxygen layer-Li layer and to permit the precipitation of the element A and the element B in triple points present in grain boundaries of the composite oxide. As a result, it is possible to further improve the safety and the charge-discharge cycle characteristics of the secondary battery.

As described above, the lithium ion secondary battery of the present invention comprising a positive electrode containing the second positive electrode

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active material permits suppressing a rapid elevation of the battery temperature when a large current flows therethrough under the state of a short circuit as in the nail sticking test, making it possible to prevent in advance rupture and ignition. It follows that the safety of the secondary battery can be improved. considered reasonable to understand that the element A containing at least one element selected from the group consisting of Na, K, Ca and S and the element B containing at least one element selected from the group consisting of Si and Fe serve to lower the series resistance of the positive electrode active material so as to produce a synergetic effect that the heat generation caused by the Joule heat derived from the flow of a large current can be suppressed, thereby producing the prominent effect described above. Also, the secondary battery of the present invention makes it possible to realize a large discharge capacity, excellent cycle characteristics and excellent large current discharge characteristics (discharge rate characteristics).

(Third Positive Electrode Active material)

The third positive electrode active material used in the present invention comprises a composite oxide having a composition represented by chemical formula (3) given below:

$$\text{Li}_{x}(\text{Ni}_{1-v-s}\text{Co}_{v}\text{Me2}_{s})(\text{O}_{2-z}\text{X}_{z}) + \text{A}$$
 ... (3)

where Me2 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, v, s and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 0.5$, $0.005 \le s \le 0.5$ and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm.

The molar ratio x of lithium is defined to fall within a range of between 0.02 and 1.3 as described above. If the molar ratio x of lithium is lower than 0.02, the crystal structure of the composite oxide is rendered highly unstable, with the result that the cycle characteristics of the secondary battery are degraded and the safety of secondary battery is lowered. On the other hand, if the molar ratio x of lithium exceeds 1.3, the discharge capacity and the safety of the secondary battery are lowered. It is more desirable for the molar ratio x of lithium to fall within a range of between 0.05 and 1.2.

The composite oxide also contains Co. Where Co is contained in the composite oxide, it is possible to suppress the spurting of gas from within the secondary

battery in the event of a short circuit as in the nail sticking test and to further suppress the temperature elevation of the secondary battery in the event of a short circuit. It is desirable for the molar ratio v of Co to fall within a range of between 0.005 and 0.5. If the molar ratio v of Co is lower than 0.005, it is difficult to improve sufficiently the safety of the secondary battery. On the other hand, if the molar ratio v of Co exceeds 0.5, the charge-discharge cycle characteristics and the discharge capacity of the secondary battery are markedly lowered.

The composite oxide also contains the element Me2. It is desirable for the molar ratio s of the element Me2 to fall within a range of between 0.005 and 0.5. If the molar ratio s of the element Me2 is lower than 0.005, it is difficult to improve sufficiently the safety of the secondary battery. On the other hand, if the molar ratio s of the element Me2 exceeds 0.5, the discharge capacity of the secondary battery is markedly lowered. It is more desirable for the molar ratio s of the element Me2 to fall within a range of between 0.01 and 0.35. Also, it is desirable to use Al, Ti, Mn, Nb or Ta as the element Me2.

The composite oxide also contains the halogen
25 element X. It is desirable for the molar ratio z of
the halogen element X to fall within a range of between
0.01 and 0.5. If the molar ratio z of the halogen

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element X is lower than 0.01, it is difficult to improve the cycle characteristics and the safety of the secondary battery. On the other hand, if the molar ratio z of the halogen element X exceeds 0.5, the discharge capacity of the secondary battery is markedly lowered. It is more desirable for the molar ratio z of the halogen element X to fall within a range of between 0.02 and 0.3. Also, it is desirable to use F as the halogen element X.

In one of the desired compositions of the composite oxide used in the present invention, the halogen element X contains F, the molar ratio x falls within a range of between 0.05 and 1.2, the molar ratio v falls within a range of between 0.005 and 0.5, the molar ratio s falls within a range of between 0.01 and 0.35, and the molar ratio z falls within a range of between 0.02 and 0.3.

Where Na is contained in the composite oxide, the Na content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Na content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where K is contained in the composite oxide, the K

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content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the K content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where S is contained in the composite oxide, the S content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the S content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

It is desirable for the element A to include Ca in addition to at least one kind of the element selected from the group consisting of Na, K and S. It is desirable for the Ca content of the composite oxide to be not higher than 500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Ca content to fall within a range of between 20 ppm and 500 ppm, furthermore desirably between 50 ppm and 500 ppm.

It is possible to improve sufficiently the safety of the secondary battery even if the elements A are added singly. However, it is possible to further improve the safety of the secondary battery by adding a plurality of kinds of the elements A in combination. The desirable combinations of the elements A include the combinations described previously in conjunction with the first positive electrode active material.

It is desirable for the total content "a" of the elements A in the composite oxide to fall within a range of between 600 ppm and 7000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the total content "a" of the elements A to fall within a range of between 1000 ppm and 5000 ppm.

In the composite oxide represented by the chemical formula (3), it is desirable for at least a part of the element A containing at least one element selected from the group consisting of Na, K, S and Ca to be segregated. Particularly, it is desirable for at least a part of the element A to be precipitated in the triple point present in the boundary of the crystal grains of the composite oxide. If the crystal structure of the composite oxide is constructed to meet the particular requirement, it is possible to further improve both the safety and the cycle characteristics

of the secondary battery.

The composite oxide used in the present invention can be manufactured by, for example, a solid state reaction process, a coprecipitation process, or hydrothermal synthesis. Particularly, it is desirable to obtain the composite oxide having a composition represented by chemical formula (3) by firing a powdery mixture of the compounds of each element at 450 to 550° C for 2 to 20 hours with an oxygen gas flow, followed by further firing the mixture at 630 to 730° C for 2 to 50 hours with an oxygen gas flow. employing the particular process, it is possible to arrange regularly the structure of the oxygen layer-Li layer-oxygen layer-(Ni+Co+Me2) layer-oxygen layer-Li layer and to permit the precipitation of the element A in triple points present in grain boundaries of the composite oxide. As a result, it is possible to further improve the safety and the charge-discharge cycle characteristics of the secondary battery.

As described above, the lithium ion secondary battery of the present invention comprising a positive electrode containing the third positive electrode active material permits effectively suppressing a rapid elevation of the battery temperature when a large current flows therethrough under the state of a short circuit as in the nail sticking test, making it possible to prevent in advance dangers such as rupture

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and ignition. It follows that the safety of the secondary battery can be markedly improved. Also, the secondary battery of the present invention makes it possible to obtain the large discharge capacity, and to further improve the cycle characteristics and the large current discharge characteristics (discharge rate characteristics).

(Fourth Positive Electrode Active material)

The fourth positive electrode active material contains a composite oxide having a composition represented by chemical formula (4) given below:

 $\text{Li}_{X}(\text{Ni}_{1-v-s}\text{Co}_{v}\text{Me2}_{s})(\text{O}_{2-z}\text{X}_{z}) + \text{A} + \text{bB}$... (4)

where Me2 is at least one kind of an element selected from the group consisting of B, Mg, Al, Sc, Ti, V, Cr, Mn, Cu, Zn, Ga, Y, Zr, Nb, Mo, Tc, Ru, Sn, La, Hf, Ta, W, Re, Pb, and Bi, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, v, s and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 0.5$, $0.005 \le s \le 0.5$ and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm, B contains at least one element selected from the group consisting of Si and Fe, and the content b of the element B in said composite oxide falls within a range of between 20 ppm and 500 ppm.

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The molar ratio x of lithium is defined to fall within a range of between 0.02 and 1.3 as described above for the same reasons as those described previously in conjunction with the third positive electrode active material. It is more desirable for the molar ratio x of lithium to fall within a range of between 0.05 and 1.2.

The composite oxide also contains Co. It is desirable for the molar ratio v of Co to fall within a range of between 0.005 and 0.5 for the same reasons as those described previously in conjunction with the third positive electrode active material.

The composite oxide also contains the element Me2. It is desirable for the molar ratio s of the element Me2 to fall within a range of between 0.005 and 0.5 for the same reasons as those described previously in conjunction with the third positive electrode active material. It is more desirable for the molar ratio s of the element Me2 to fall within a range of between 0.01 and 0.35. Also, it is desirable to use the elements described previously in conjunction with the third positive electrode active material as the element Me2.

The composite oxide also contains the halogen
25 element X. It is desirable for the molar ratio z of
the halogen element X to fall within a range of between
0.01 and 0.5 for the same reasons as those described

previously in conjunction with the third positive electrode active material. It is more desirable for the molar ratio z of the halogen element X to fall within a range of between 0.02 and 0.3. Also, it is desirable to use F as the halogen element X.

Where Na is contained in the composite oxide, the Na content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Na content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where K is contained in the composite oxide, the K content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the K content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where S is contained in the composite oxide, the S content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive

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electrode active material. It is more desirable for the S content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

It is desirable for the element A to include Ca in addition to at least one kind of the element selected from the group consisting of Na, K and S. It is desirable for the Ca content of the composite oxide to be not higher than 500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Ca content to fall within a range of between 20 ppm and 500 ppm, furthermore desirably between 50 ppm and 500 ppm.

It is possible to improve sufficiently the safety of the secondary battery even if the elements A are added singly. However, it is possible to further improve the safety of the secondary battery by adding a plurality of kinds of the elements A in combination. The desirable combinations of the elements A include the combinations described previously in conjunction with the first positive electrode active material.

It is desirable for the total content "a" of the elements A in the composite oxide to fall within a range of between 600 ppm and 7000 ppm for the same reasons as those described previously in conjunction

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with the first positive electrode active material. It is more desirable for the total content "a" of the elements A to fall within a range of between 1000 ppm and 5000 ppm.

The content b of element B in the composite oxide is defined to fall within the range described above for the same reasons as those described previously in conjunction with the first positive oxide active material. It is more desirable for the content b of element B to fall within a range of between 20 ppm and 250 ppm. Further, where the element B contains both of Si and Fe, it is possible to further improve the safety and the charge-discharge cycle characteristics of the secondary battery.

In the composite oxide represented by the chemical formula (4), it is desirable for at least a part of at least one kind of the element A selected from the group consisting of Na, K, S and Ca and for at least a part of at least one kind of the element B selected from the group consisting of Si and Fe to be segregated.

Particularly, it is desirable for at least a part of one of the element A and the element B to be precipitated in triple points present in grain boundaries of the composite oxide. If the crystal structure of the composite oxide is constructed to meet the particular requirement, it is possible to further improve both the safety and the cycle characteristics

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of the secondary battery.

The composite oxide used in the present invention can be manufactured by, for example, a solid state reaction process, a coprecipitation process, or hydrothermal synthesis. Particularly, it is desirable to obtain the composite oxide having a composition represented by chemical formula (4) by firing a powdery mixture of the compounds of each element at 450 to 550 $^{\circ}$ C for 2 to 20 hours with an oxygen gas flow, followed by further firing the mixture at 630 to 730° C for 2 to 50 hours with an oxygen gas flow. By employing the particular process, it is possible to arrange regularly the structure of the oxygen layer-Li layer-oxygen layer-(Ni+Co+Me2) layer-oxygen layer-Li layer and to permit the precipitation of the element A and the element B in triple points present in grain boundaries of the composite oxide. As a result, it is possible to further improve the safety and the chargedischarge cycle characteristics of the secondary battery.

As described above, the lithium ion secondary battery of the present invention comprising a positive electrode containing the fourth positive electrode active material permits effectively suppressing a rapid elevation of the battery temperature when a large current flows therethrough under the state of a short circuit as in the nail sticking test, making it

possible to prevent in advance dangers such as rupture and ignition. It follows that the safety of the secondary battery can be markedly improved. Also, the secondary battery of the present invention makes it possible to obtain a large discharge capacity and to further improve the charge-discharge cycle characteristics and the large current discharge characteristics (discharge rate characteristics).

(Fifth Positive Electrode Active material)

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The fifth positive electrode active material contains a composite oxide having a composition represented by chemical formula (5) given below:

 $\text{Li}_{\mathbf{X}}(\text{Ni}_{1-\mathbf{V}-\mathbf{t}}\text{Co}_{\mathbf{V}}\text{Me3}_{\mathbf{t}}) (\text{O}_{2-\mathbf{Z}}\text{X}_{\mathbf{Z}}) + A \qquad \dots (5)$

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where Me3 is at least one kind of an element selected from the group consisting of Ti, V, Cr, Zr, Nb, Mo, Hf, Ta and W, X is at least one kind of a halogen element selected from the group consisting of F, Cl, Br and I, the molar ratios x, v, t and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 1.3$, $0.005 \le t \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, and each of the Na content, the K content and the S content of the composite oxide falls within a range of between

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The molar ratio x of lithium falls within a range of between 0.02 and 1.3 as described above. If the molar ratio x of lithium is lower than 0.02, the

600 ppm and 3,000 ppm.

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crystal structure of the composite oxide is rendered highly unstable so as to degraded the cycle characteristics of the secondary battery and to lower the safety of the secondary battery. On the other hand, if the molar ratio x of lithium exceeds 1.3, the discharge capacity and the safety of the secondary battery are lowered. It is more desirable for the molar ratio x of lithium to fall within a range of between 0.05 and 1.2.

The molar ratio t of the element Me3 falls within a range of between 0.005 and 0.5 as described above. If the molar ratio t of the element Me3 is lower than 0.005, it is difficult to improve the safety of the secondary battery. On the other hand, if the molar ratio t of the element Me3 exceeds 0.5, the discharge capacity of the secondary battery is markedly lowered. It is more desirable for the molar ratio t of the element Me3 to fall within a range of between 0.01 and 0.35. Also, it is desirable to use Ti, Nb or Ta as the element Me3.

The composite oxide also contains Co. Where Co is contained in the composite oxide, it is possible to suppress the spurting of the gas from within the secondary battery in the event of a short circuit as in the nail sticking test and to further suppress the temperature elevation of the secondary battery in the event of a short circuit. It is desirable for the

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molar ratio v of Co to fall within a range of between 0.005 and 0.5. If the molar ratio v of Co is lower than 0.005, it is difficult to improve sufficiently the safety of the secondary battery. On the other hand, if the molar ratio v of Co exceeds 0.5, the charge-discharge cycle characteristics and the discharge capacity of the secondary battery are markedly lowered.

The composite oxide also contains the halogen element X. It is desirable for the molar ratio z of the halogen element X to fall within a range of between 0.01 and 0.5. If the molar ratio z of the halogen element X is lower than 0.01, it is difficult to improve the cycle characteristics and the safety of the secondary battery. On the other hand, if the molar ratio z of the halogen element X exceeds 0.5, the discharge capacity of the secondary battery is markedly lowered. It is more desirable for the molar ratio z of the halogen element X to fall within a range of between 0.02 and 0.3. Also, it is desirable to use F as the halogen element X.

In one of the desired compositions of the composite oxide used in the present invention, the halogen element X contains F, the molar ratio x falls within a range of between 0.05 and 1.2, the molar ratio v falls within a range of between 0.005 and 0.5, the molar ratio t falls within a range of between 0.01 and

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0.35, and the molar ratio z falls within a range of between 0.02 and 0.3.

Where Na is contained in the composite oxide, the Na content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Na content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where K is contained in the composite oxide, the K content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the K content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where S is contained in the composite oxide, the S content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the S content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those

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described previously in conjunction with the first positive electrode active material.

It is desirable for the element A to include Ca in addition to at least one kind of the element selected from the group consisting of Na, K and S. It is desirable for the Ca content of the composite oxide to be not higher than 500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Ca content to fall within a range of between 20 ppm and 500 ppm, furthermore desirably between 50 ppm and 500 ppm.

It is possible to improve sufficiently the safety of the secondary battery even if the elements A are added singly. However, it is possible to further improve the safety of the secondary battery by adding a plurality of kinds of the elements A in combination. The desirable combinations of the elements A include the combinations described previously in conjunction with the first positive electrode active material.

It is desirable for the total content "a" of the elements A in the composite oxide to fall within a range of between 600 ppm and 7000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

It is more desirable for the total content "a" of the elements A to fall within a range of between 1000 ppm

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and 5000 ppm.

In the composite oxide represented by the chemical formula (5), it is desirable for at least a part of the element A containing at least one element selected from the group consisting of Na, K, S and Ca to be segregated. Particularly, it is desirable for at least a part of the element A to be precipitated in triple points present in grain boundaries of the composite oxide. If the crystal structure of the composite oxide is constructed to meet the particular requirement, it is possible to further improve both the safety and the cycle characteristics of the secondary battery.

The composite oxide used in the present invention can be manufactured by, for example, a solid state reaction process, a coprecipitation process, or hydrothermal synthesis. Particularly, it is desirable to obtain the composite oxide having a composition represented by chemical formula (5) by firing a powdery mixture of the compounds of each element at 450 to 550°C for 2 to 20 hours with an oxygen gas flow, followed by further firing the mixture at 630 to 730°C for 2 to 50 hours with an oxygen gas flow. By employing the particular process, it is possible to arrange regularly the structure of the oxygen layer-Li layer-oxygen layer-(Ni+Co+Me3) layer-oxygen layer-Li layer and to permit the precipitation of the element A in triple points present in grain boundaries of the

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composite oxide. As a result, it is possible to further improve the safety and the charge-discharge cycle characteristics of the secondary battery.

As described above, the lithium ion secondary battery of the present invention comprising a positive electrode containing the fifth positive electrode active material permits effectively suppressing a rapid elevation of the battery temperature when a large current flows therethrough under the state of a short circuit as in the nail sticking test, making it possible to prevent in advance dangers such as rupture and ignition. It follows that the safety of the secondary battery can be markedly improved. Also, the secondary battery of the present invention makes it possible to obtain a large discharge capacity and to further improve the charge-discharge cycle characteristics and the large current discharge characteristics (discharge rate characteristics).

(Sixth Positive Electrode Active material)

The sixth positive electrode active material contains a composite oxide having a composition represented by chemical formula (6) given below:

 $\text{Li}_{X}(\text{Ni}_{1-v-t}\text{Co}_{v}\text{Me3}_{t}) (O_{2-z}X_{z}) + A + bB \dots$ (6)

where Me3 is at least one kind of an element selected from the group consisting of Ti, V, Cr, Zr, Nb, Mo, Hf, Ta and W, X is at least one kind of a halogen element selected from the group consisting of

F, Cl, Br and I, the molar ratios x, v, t and z are $0.02 \le x \le 1.3$, $0.005 \le v \le 0.5$, $0.005 \le t \le 0.5$, and $0.01 \le z \le 0.5$, A contains at least one element selected from the group consisting of Na, K and S, each of the Na content, the K content and the S content of the composite oxide falls within a range of between 600 ppm and 3,000 ppm, B consists essentially of at least one kind of the element selected from the group consisting of Si and Fe, and the content b of the element B in said composite oxide falls within a range of between 20 ppm and 500 ppm.

The molar ratio x of lithium falls within a range of between 0.02 and 1.3 as described above for the same reasons as those described previously in conjunction with the fifth positive electrode active material. It is more desirable for the molar ratio x of lithium to fall within a range of between 0.05 and 1.2.

The molar ratio t of the element Me3 falls within a range of between 0.005 and 0.5 as described above for the same reasons as those described previously in conjunction with the fifth positive electrode active material. It is more desirable for the molar ratio t of the element Me3 to fall within a range of between 0.01 and 0.35. Also, it is desirable to use the elements described previously in conjunction with the fifth positive electrode active material as the element Me3.

The composite oxide also contains Co. It is desirable for the molar ratio v of Co to fall within a range of between 0.005 and 0.5 for the same reasons as those described previously in conjunction with the fifth positive electrode active material.

The composite oxide also contains the halogen element X. It is desirable for the molar ratio z of the halogen element X to fall within a range of between 0.01 and 0.5 for the same reasons as those described previously in conjunction with the fifth positive electrode active material. It is more desirable for the molar ratio z of the halogen element X to fall within a range of between 0.02 and 0.3. Also, it is desirable to use F as the halogen element X.

Where Na is contained in the composite oxide, the Na content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Na content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where K is contained in the composite oxide, the K content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive

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electrode active material. It is more desirable for the K content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

Where S is contained in the composite oxide, the S content should fall within a range of between 600 ppm and 3,000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the S content to fall within a range of between 1,000 ppm and 2,500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material.

It is desirable for the element A to include Ca in addition to at least one kind of the element selected from the group consisting of Na, K and S. It is desirable for the Ca content of the composite oxide to be not higher than 500 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the Ca content to fall within a range of between 20 ppm and 500 ppm, furthermore desirably between 50 ppm and 500 ppm.

It is possible to improve sufficiently the safety of the secondary battery even if the elements A are added singly. However, it is possible to further

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improve the safety of the secondary battery by adding a plurality of kinds of the elements A in combination. The desirable combinations of the elements A include the combinations described previously in conjunction with the first positive electrode active material.

It is desirable for the total content "a" of the elements A in the composite oxide to fall within a range of between 600 ppm and 7000 ppm for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the total content "a" of the elements A to fall within a range of between 1000 ppm and 5000 ppm.

It is desirable for the content b of the element B in the composite oxide to fall within the range described above for the same reasons as those described previously in conjunction with the first positive electrode active material. It is more desirable for the content b of the element B to fall within a range of between 20 ppm and 250 ppm. Also, it is possible to further improve the safety and the charge-discharge cycle characteristics of the secondary battery by using Si and Fe as the element B.

In the composite oxide represented by the chemical formula (6), it is desirable for at least a part of the element A containing at least one element selected from the group consisting of Na, K, S and Ca and for at

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least a part of the element B containing at least one element selected from the group consisting of Si and Fe to be segregated. Particularly, it is desirable for at least a part of at least one of the element A and the element B to be precipitated in the triple point present in the boundary of the crystal grains of the composite oxide. If the crystal structure of the composite oxide is constructed to meet the particular requirement, it is possible to further improve both the safety and the cycle characteristics of the secondary battery.

The composite oxide used in the present invention can be manufactured by, for example, a solid state reaction process, a coprecipitation process, or hydrothermal synthesis. Particularly, it is desirable to obtain the composite oxide having a composition represented by chemical formula (6) by firing a powdery mixture of the compounds of each element at 450 to 550° C for 2 to 20 hours with an oxygen gas flow, followed by further firing the mixture at 630 to 730° C for 2 to 50 hours with an oxygen gas flow. By employing the particular method, it is possible to arrange regularly the structure of the oxygen layer-Li layer-oxygen layer-(Ni+Co+Me3) layer-oxygen layer-Li layer and to permit the precipitation of the element A and the element B in triple points present in grain boundaries of the composite oxide. As a result, it is

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possible to further improve the safety and the chargedischarge cycle characteristics of the secondary battery.

As described above, the lithium ion secondary battery of the present invention comprising a positive electrode containing the sixth positive electrode active material permits effectively suppressing a rapid elevation of the battery temperature when a large current flows therethrough under the state of a short circuit as in the nail sticking test, making it possible to prevent in advance dangers such as rupture and ignition. It follows that the safety of the secondary battery can be markedly improved. Also, the secondary battery of the present invention makes it possible to obtain a large discharge capacity and to further improve the charge-discharge cycle characteristics and the large current discharge characteristics (discharge rate characteristics).

Among the first to sixth positive electrode active materials described above, it is desirable to use the first to fourth positive electrode active materials because these positive electrode active materials permit improving both the safety and the cycle characteristics. Particularly, it is desirable to use the third and fourth positive electrode active materials.

The positive electrode is prepared by mixing at

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least one kind of the first to sixth positive electrode active materials described above, an electrical conduction aid and a binder, followed by pressing the resultant mixture against a current collector. Also, the positive electrode is prepared by suspending the positive electrode active material noted above, the electrical conduction aid and the binder in a suitable solvent, followed by coating a current collector with the resultant suspension and subsequently drying the coated suspension.

The electrical conduction aid used in the present invention includes, for example, acetylene black, carbon black and graphite.

The binder used in the present invention includes, for example, polytetrafluoro ethylene (PTFE), polyvinylidene fluoride (PVdF), ethylene-propylene-diene copolymer (EPDM), and styrene-butadiene rubber (SBR).

Concerning the mixing amounts of the positive electrode active material, the electrical conduction aid and the binder, it is desirable for the mixing amount of the positive electrode active material to fall within a range of between 80% and 95% by weight, it is desirable for the mixing amount of the electrical conduction aid to fall within a range of between 3% and 20% by weight, and it is desirable for the mixing amount of the binder to fall within a range of between

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2% and 7% by weight. It is possible to use a conductive substrate of a porous structure or a conductive substrate of a nonporous structure as the current collector. Also, it is possible for the current collector to be formed of, for example, aluminum, stainless steel or nickel.

2) Separator

It is possible to use, for example, a synthetic resin unwoven fabric, a polyethylene porous film, or a polypropylene porous film as the separator.

3) Negative Electrode

The negative electrode includes a material capable of absorbing (doping)/releasing (desorbing) lithium. The particular material included in the negative electrode includes, for example, a lithium metal, a Li-containing alloy capable of absorbing/desorbing lithium, a metal oxide capable of absorbing/desorbing lithium, a metal sulfide capable of absorbing/desorbing lithium, a metal nitride capable of absorbing/desorbing lithium, a chalcogen compound capable of absorbing desorbing lithium, and a carbonaceous material capable of absorbing desorbing lithium ions. Particularly, it is desirable to use a negative electrode containing the chalcogen compound or the carbonaceous material noted above, because the negative electrode containing the particular compound is highly safe and permits improving the cycle life of

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the secondary battery.

The carbonaceous material capable of absorbing/desorbing lithium ions includes, for example, coke, a carbon fiber, a vapor-grown-carbon material, graphite, a resin calcined body, a mesophase pitch based carbon fiber, and a mesophase based spherical carbon. It is desirable to use the carbonaceous materials noted above because the particular carbonaceous materials permit improving the electrode capacity.

The chalcogen compounds used in the present invention include, for example, titanium disulfide, molybdenum disulfide, niobium selenide, and tin oxide. If these chalcogen compounds are contained in the negative electrode, the battery voltage is lowered. However, since the capacity of the negative electrode is increased, it is possible to increase the capacity of the secondary battery.

The negative electrode containing the carbonaceous material can be prepared by kneading a mixture of the carbonaceous material and the binder noted above in the presence of a solvent, followed by coating a current collector with the resultant kneaded material and subsequently drying the coated material.

The binder used for preparation of the kneaded material noted above includes, for example, polytetrafluoro ethylene (PTFE), polyvinylidene

fluoride (PVdF), ethylene-propylene-diene copolymer (EPDM), and styrene-butadiene rubber (SBR). Concerning the mixing amounts of the carbonaceous material and the binder, it is desirable for the mixing amount of the carbonaceous material to fall within a range of between 90% and 98% by weight, and it is desirable for the mixing amount of the binder to fall within a range of between 2% and 10% by weight. Also, it is possible to use a conductive substrate made of, for example, copper, stainless steel or nickel as the current collector. It is possible for the current collector to be either porous or nonporous.

4) Liquid Nonaqueous Electrolyte

The liquid nonaqueous electrolyte is prepared by dissolving a solute in a nonaqueous solvent. The nonaqueous solvent used in the present invention includes, for example, a cyclic carbonate, a straight chain carbonates such as ethylene carbonate, propylene carbonate, diethyl carbonate, dimethyl carbonate or methyl ethyl carbonate, a cyclic ether, a straight chain ether such as 1,2-dimethoxy ethane, or 2-methyl tetrahydro furan, a cyclic ester, and a straight chain ester such as γ -butyrolactone, γ -valerolactone, σ -valerolactone, methyl acetate, ethyl acetate, propyl acetate, isopropyl acetate, methyl propionate, ethyl propionate or propyl propionate. These nonaqueous solvents can be used singly or in

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the form of a mixed solvent prepared by mixing 2 to 5 kinds of the nonaqueous solvents exemplified above, though the nonaqueous solvents used in the present invention are not limited to the compounds exemplified above.

The solute used in the present invention includes, for example, lithium salts such as lithium perchlorate (LiClO_4), lithium hexafluoro phosphate (LiPF_6), lithium tetrafuluoro borate (LiBF_4), lithium hexafluoro arsenate (LiAsF_6), lithium trifluoro meta-sulfonate (LiCF_3SO_3), and bis-trifluoromethyl sulfonyl imide lithium [$\text{LiN}(\text{CF}_3\text{SO}_2)_2$]. These solutes can be used singly or in the form of a mixture of two or three kinds of these lithium salts, though the solutes used in the present invention for preparing the liquid nonaqueous electrolyte are not limited to the compounds exemplified above.

It is desirable for the amount of the solute dissolved in the nonaqueous solvent to fall within a range of between 0.5 and 2 mol/L.

FIGS. 1 to 3 collectively show as an example the construction of the lithium ion secondary battery of the present invention. Specifically, FIG. 1 is an oblique view, partly broken away, showing the construction of a cylindrical lithium ion secondary battery as an example of the lithium ion secondary battery of the present invention. FIG. 2 is

a cross-sectional view showing a thin lithium ion secondary battery as an example of the lithium ion secondary battery of the present invention. Further, FIG. 3 is a cross-sectional view showing in a magnified fashion the portion A of the lithium ion secondary battery shown in FIG. 2.

As shown in FIG. 1, an insulating body 2 is arranged in the bottom portion of a cylindrical container 1 made of, for example, stainless steel. An electrode group 3 is arranged within the container 1. The electrode group 3 is prepared by spirally winding a band-like material prepared by laminating a positive electrode 4, a separator 5 and a negative electrode 6 in the order mentioned.

A nonaqueous electrolyte is housed in the container 1. A PTC element 7 having a hole formed in the central portion, a safety valve 8 arranged on the PTC element 7 and a hat-shaped positive electrode terminal 9 arranged on the safety valve 8 are fixed by caulking to the upper open portion of the container 1 with an insulating gasket 10 interposed therebetween. Incidentally, a safety mechanism acting as a gas releasing hole (not shown) is incorporated in the positive electrode terminal 9. A positive electrode lead 11 is connected at one end to the positive electrode 4 and to the PTC element 7 at the other end. Also, the negative electrode 6 is connected via a

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negative electrode lead (not shown) to the container 1 acting as a negative electrode terminal.

As shown in FIG. 2, an electrode group 22 is housed in a housing container 21 made of, for example, a film material. The film material used in the present invention includes, for example, a metal film, a resin sheet such as a thermoplastic resin sheet, and a sheet comprising a metal layer and a resin layer such as a thermoplastic resin layer formed on one surface or both surfaces of the metal layer. The electrode group 22 is prepared by winding flat a laminate structure comprising a positive electrode, a separator and a negative electrode. The laminate structure comprises a separator 23; a positive electrode 26 including a positive electrode layer 24, a positive electrode current collector 25 and a positive electrode layer 24; a separator 23; a negative electrode 29 including a negative electrode layer 27, a negative electrode current collector 28, and a negative electrode layer 27; a separator 23; a positive electrode 26 including a positive electrode layer 24, a positive electrode current collector 25 and a positive electrode layer 24; a separator 23; and a negative electrode 29 including a negative electrode layer 27, and a negative electrode current collector 28, which are laminated in the order mentioned as viewed from the lower end. The negative electrode current collector 28 is

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positioned to constitute the outermost circumferential surface of the electrode group 22. A band-like positive electrode lead 30 is connected at one end to the positive electrode current collector 25 and is withdrawn at the other end portion from within the housing container 21. On the other hand, a band-like negative electrode lead 31 is connected at one end portion to the negative electrode current collector 28 of the electrode group 22 and is withdrawn at the other end portion from within the housing container 21.

Some Examples of the present invention will now be described with reference to the accompanying drawings.

Examples 1 to 26:

Prepared as starting materials were powders of LiOH· H_2O , Ni(OH)₂, oxides, carbonates and nitrates of the element Mel, NaOH, KOH, Ca(OH)₂, sodium sulfide (Na₂S· $9H_2O$) as a sulfide compound, and a sulfate compound (NiSO₄· $6H_2O$). These powdery compounds were selected to form the composition shown in Tables 1 and 2, i.e., Li_{1.1}(Ni_{0.88}Mel_{0.02})(O_{1.9}X_{0.1}) + aA, and mixed, followed by further mixing the composition in a Henschel mixer for 30 minutes so as to prepare a mixed powder. The mixed powder was put in an alumina sagger for firing. Firing was performed at 480°C for 10 hours while allowing an oxygen gas to flow at a rate of 5 liter/min, followed by further firing the mixed powder at 700°C for 20 hours with an oxygen gas flow at

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a rate of 5 liter/min so as to obtain a positive electrode active material.

Incidentally, the composition of the positive electrode active material was measured by glow discharge mass spectrometry (GDMS). In the GDMS method, several kilovolts (kV) of voltage is applied to one electrode used as a sample in an Ar gas atmosphere of about 1 Torr so as to form a glow discharge and to apply a sputtering to the surface of the sample. sample ions thus formed are withdrawn through an aperture formed in the electrode and are accelerated so as to perform the mass analysis. The content of each of the elements forming the composite oxide is obtained by this glow discharge type mass spectrometry, and the content of each of the elements other than the element A is converted into the molar percentage (mol %) so as to obtain the chemical formula. The composition analysis was performed by the glow discharge mass spectrometry in the positive electrode active materials obtained in the Examples described in the following.

A positive electrode composition consisting of 92.2% by weight of the positive electrode active material, 1.8% by weight of an acetylene black, 2.2% by weight of a synthetic graphite, and 3.8% by weight of polyvinylidene fluoride was prepared by adding a lithium-containing composite oxide powder used as a positive electrode active material, and an electrical

conduction agent formed of an acetylene black and a synthetic graphite in a solution prepared by dissolving polyvinylidene fluoride in N-methyl-2-pyrrolidone while stirring the system. Both surfaces of an aluminum foil having a thickness of 20 μ m were coated with the positive electrode composition thus prepared, followed by drying the coating and subsequently pressing the coating by using a roller press machine.

<Preparation of Negative Electrode>

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A carbonaceous material was prepared by carbonizing mesophase pitch carbon fibers, which were prepared by using mesophase pitch as the raw material, at 1,000°C in an argon gas atmosphere, followed by appropriately pulverizing the carbonized mesophase pitch carbon fibers such that the pulverized fibers had an average fiber length of 30 $\mu \rm m$ and an average fiber diameter of 11 $\mu \rm m$, and that the amount of the particles having a particle diameter not larger than 0.5 $\mu \rm m$ is decreased (not larger than 5%) and the particles having the particle diameters falling within a range of between 1 $\mu \rm m$ and 80 $\mu \rm m$ to occupy 90% by volume of the pulverized fibers, and subsequently graphitizing the pulverized fibers at 3,000°C in an argon gas atmosphere.

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A negative electrode composition consisting of 86.5% by weight of the carbonaceous material, 9.5% by weight of a synthetic graphite, and 4% by weight of

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polyvinylidene fluoride was prepared by adding the carbonaceous material thus prepared and a synthetic graphite to a solution prepared by dissolving polyvinylidene fluoride in N-methyl-2-pyrrolidone.

Then, both surfaces of a copper foil having a thickness of 15 μm were coated with the negative electrode composition thus prepared, followed by drying the coating and subsequently pressing the coating by using a roller press machine so as to prepare the negative electrode. In preparing the positive electrode and the negative electrode, the loading density and the electrode length were adjusted to permit a ratio (capacity balance) of the design capacity of the negative electrode to the design capacity of the positive electrode after pressing to fall within a range of between 1.03 and 1.1.

A nonaqueous electrolyte was prepared by dissolving lithium hexafluoro phosphate (LiPF $_6$) in a mixed nonaqueous solvent prepared by mixing ethylene carbonate (EC) and methyl ethyl carbonate (MEC) at a volume ratio of 1 : 1.

<Assembly of Battery>

A positive electrode lead made of aluminum and a negative electrode lead made of nickel were welded to the positive electrode and the negative electrode,

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respectively, followed by laminating the positive electrode, a separator formed of a polyethylene porous film and the negative electrode in the order mentioned. The laminate structure thus prepared spirally wound such that the negative electrode is positioned to form the outer surface, thereby preparing an electrode group.

The electrode group thus prepared was housed in a cylindrical container having a bottom. Then, the negative electrode lead was welded to the bottom portion of the cylindrical container, and the positive electrode lead was welded to the safety valve arranged in the open portion of the cylindrical container.

Further, 4 mL of the nonaqueous electrolyte was poured into the cylindrical container so as to permit the electrode group to be impregnated sufficiently with the nonaqueous electrolyte. Still further, the positive electrode terminal was arranged on the safety valve and fixed by the caulking, thereby assembling a cylindrical lithium ion secondary battery (18650 size) having a rated design capacity of 1,600 mAh.

Comparative Examples 1 to 9:

Prepared as starting materials were powders of LiOH· H_2O , Ni(OH)₂, oxides, carbonates and nitrates of the element Me1, NaOH, KOH, Ca(OH)₂, sodium sulfide (Na₂S· $9H_2O$) as a sulfide compound, and a sulfate compound (NiSO₄· $6H_2O$). These powdery compounds were

selected to form the composition shown in Table 3 and mixed, followed by further mixing the composition in a Henschel mixer for 30 minutes so as to prepare a mixed powder. The mixed powder was put in an alumina sagger for firing. Firing was performed at 480° C for 10 hours while allowing an oxygen gas to flow at a rate of 5 liter/min, followed by further firing the mixed powder at 700° C for 20 hours with an oxygen gas flow at a rate of 5 liter/min so as to obtain a positive electrode active material.

A cylindrical lithium ion secondary battery was prepared as in Example 1, except that the positive electrode active material thus prepared was used.

<Nail Sticking Test>

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A nail sticking test was applied to the secondary battery prepared in each of Examples 1 to 26 and Comparative Examples 1 to 9. In the first step, the secondary battery was charged. The charging was performed to reach 4.2 V with a current value corresponding to 0.2C, which is calculated on the basis of the rated design capacity of the secondary battery, followed by maintaining the constant voltage of 4.2 V. The charging was applied for 8 hours in total. After the secondary battery was charged to 4.2 V, a safety of the secondary battery was studied by a nail sticking test. The nail used in the test had a diameter of 2 mm and the nail sticking speed was 135 mm/sec. Also, the

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temperature elevation of the secondary battery in the nail sticking test was measured by a thermocouple mounted on the outer surface of the secondary battery. Tables 1 to 3 show the situation in respect of the occurrence or nonoccurrence of rupture/ignition caused by the nail sticking test and the battery temperature in the nail sticking test.

<Initial Capacity and Charge-Discharge Cycle Life> A charge-discharge cycle test was applied at room temperature to the secondary battery prepared in each of Examples 1 to 26 and Comparative Examples 1 to 9 so as to determine the discharge capacity (initial discharge capacity) after the first cycle and the reduction rate of the discharge capacity after 300 cycles. Tables 1 to 3 show the results. charge-discharge cycle test, the secondary battery was charged to 4.2 V under a current corresponding to 0.5C of the rated design capacity, followed by maintaining the constant voltage of 4.2 V. The charging was performed for 5 hours in total. On the other hand, the secondary battery was discharged to 2.7 V under the same current value. A rest time of 30 minutes was provided between the charging and the discharging.

<High Rate Discharge Characteristics (Discharge
Rate Characteristics)>

The secondary battery prepared in each of Examples 1 to 26 and Comparative Examples 1 to 9 was

charged to 4.2 V with a current corresponding to 0.5C of the rated design capacity, followed by maintaining the constant voltage of 4.2 V. The charging was performed for 5 hours in total. Then, the secondary battery was discharged 30 minutes later to 2.7 V under a current corresponding to 5C. The discharge capacity at the time when the secondary battery was discharged under a current of 5C was compared with the initial discharge capacity measured under the conditions described previously so as to determine the rate (%) in the reduced capacity at the time of the discharge at 5C relative to the initial discharge capacity, thereby obtaining the large current discharge characteristics (discharge rate characteristics). Tables 1 to 3 also show the results.

The secondary battery prepared in each of
Examples 1 to 26 and Comparative Examples 1 to 9 was
charged to 4.2 V with a current corresponding to 0.5C
of the rated design capacity, followed by maintaining
the constant voltage of 4.2 V. The charging was
performed for 5 hours in total. Then, the secondary
battery was discharged 30 minutes later to 2.7 V under
a current corresponding to 5C. The discharge capacity
was measured after the charge-discharge operations

described above were repeated 100 times so as to determine the rate of the reduced capacity in the discharge time after 100 charge-discharge cycles relative to the initial discharge capacity, thereby obtaining the high rate cycle characteristics. Tables 1 to 3 also show the results.

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				T	Table 1			
	Li	Me1	Molar	Halogen	Molar	A	๙	Result of nail sticking
	molar		ratio		ratio		(mdd)	test
	ratio		δ	<u>.</u> .	2			
	×							
Example 1	1.10	Д	0.02	드니	0.1	Na	3000	No rupture and no ignition
-	٠ ا ٠	Ma	0.02	됴	0.1	內	3000	No rupture and no ignition
	1.10	Al	0.02	ᡏᠴ	0.1	ß	3000	No rupture and no ignition
1	1.10	SC	0.02	[±ı	0.1	Na	2500	No rupture and no ignition
1	٠ ١	>	0.02	됴	0.1	Ж	2500	No rupture and no ignition
	1.10	Cr	0.02	Ē	0.1	ß	2500	No rupture and no ignition
{	1.10	Mn	0.02	도니	0.1	Na	2000	No rupture and no ignition
1	1 10	7,n	0.02	CJ	0.1	X	2000	No rupture and no ignition
1	1 10	r. C	0.02	G7	0.1	S	2000	No rupture and no ignition
i	1 0	\$ >	0.02	C1	0.1	Na	1500	No rupture and no ignition
- 1		7.7	0.02	CJ	0.1	又	1500	No rupture and no ignition
Example 11	7 F	j N	0.02	Br	0.1	ß	1500	No rupture and no ignition
1	• •	C H	0.02	Br	0.1	Na	1000	No rupture and no ignition
1	1.10	Ru	0.02	Br	0.1	Ж	1000	No rupture and no ignition
1	' '	Sn	0.02	Br	0.1	တ	1000	No rupture and no ignition
1 -	• •	# H	0.02	 	0.1	Na	009	No rupture and no ignition
1	•	M	0.02	H	0.1	X	009	No rupture and no ignition
۳ ۲	1 -	ه ک	0.02	Н	0.1	ß	009	No rupture and no ignition
FXampre To	-	2),1						(Continued)

(Continued)

Table 1

			Table 1		
	Battery	Initial	Reduction rate	High rate	High rate cycle
	temperature	discharge	of discharge	discharge	characteristics
		capacity	capacity after	characteristics	(%)
**	sticking	[mAh]	300 cycles [%]	(%)	
	test [°C]				
Fxample 1	108	1680	8	-14	-21
ì	107	1700	-13	-16	-22
- 1	107	1750	თ 1	-15	-21
	108	1660	-10	-13	-12
	107	1700	ω !	-16	-15
1	106	1680	-14	-14	-13
	107	1690	0	-15	-14
	107	1670	<u></u> ට	-17	-14
1	107	1670	6 -	-13	-13
ļ	108	1680	-10	-14	-14
1	106	1680	-15	-16	-14
	107	1680	8	-13	-14
1	108	1660	6 1	-14	-13
	108	1670		-17	-15
1	108	1680	8	-13	-14
1	107	1680	6 -	-13	-22
	108	1710	6 1	-17	-23
1	108	1660	8	-14	-22

		Me1	Molar	Halogen	Molar	Ø	ď	Result of nail sticking test
	molar ratio	i Di	ratio V))) 1	ratio		(mdd)	
0	X 1.10	Bi	0.02	 	0.1	Na	1500	No rupture and no ignition
	•	!				Ca	300	
2.0	1.10	F.E.	0.02	[E4	0.1	Ca	500	No rupture and no ignition
						Na	1500	
						S	2000	
Example 21	1.10	Mn	0.02	ফ	0.1	S	2200	No rupture and no ignition
1						Ca	400	
						內	1300	
Example 22	1.10	QN QN	0.02	5	0.1	Ca	100	No rupture and no ignition
	1					X	1000	
						හ	1500	
e 23	1.10	C.	0.02	CI	0.1	Сa	100	No rupture and no ignition
!					-	Na	009	
e 24	1.10	Mo	0.02	Br	0.1	Na	1500	No rupture and no ignition
1						呂	1000	
						യ	2200	
Example 25	1.10	Ta	0.02	- →	0.1	內	1000	No rupture and no ignition
						Ca	50	
		-				യ	1500	
Example 26	1.10	La	0.02	দ	0.1	Na	1500	No rupture and no ignition
						G G	009	

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			Table 2		
	Battery	Initial	Reduction rate	High rate	High rate cycle
	temperature	discharge	of discharge	discharge	characteristics
	during nail	capacity	capacity after	characteristics	(%)
	sticking	[mAh]	300 cycles [8]	(%)	
	test [°C]				
Tvamole 19	107	1660	၈ ၊	6 -	6 -
Frample 10	104	1730	ı D	6 1	6 -
בס בולייוים בס	107	1720	ı	80	6 1
Examble 71	₽ O T	7			
Fxample 22	103	1770	- 7	O I	ו
	104	1690		6	-17
Exampre 70	# · ·		0	1 4	-13
Example 24	104	7 / nn	- 1	11	
Tvamrle 25	103	1770	ω Ι	-11	1
סיין סיין דייין	103	1680	-20	-33	-31
Exambre 70	001	,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,			

					Table	3		
	Li	Me1	Molar	Halogen	Molar	A	ಹ	Result of nail sticking
	molar		ratio		ratio		(mdd)	test
	ratio		X		Z			
	×							
נבו	1.10	Д	0.02	[五	0.1	Na	500	No rupture and no ignition
example 1								Oscurrence of rupture and
	1.10	В	0.02	Ĺτι	0.1	None	I	331
example 2								
	1.10	₽	0.02	[2 -1	0.1	တ	200	No rupture and no ignition
example 3								organization of rintline and
Comparative	1 10	π Ε-	0.02	Ē	0.1	None	ı	4 건 2 4
example 4	•							H-1
ו ו	1.10	>	0.02	拒	0.1	—— X	200	No rupture and no ignition
example 5								bre aritative to some
Comparative	7	>	0	Ţz	0.1	None	1	כת סו ומיס
example 6	O → →	-	•					
011: + 63 6 6 8 6 7] 			(, ,	000	Occurrence of rubture and
j ,	1.10	Zr	0.02	[1 4	- -) N	ignition
example 7			-					Occurrence of rupture and
Comparative	7	7.7	0.02	<u>F</u> 4	0.1	None	ì	
example 8) - -							
						1	7	occurrence of rupture and
L)	1.10	Zr	0.02	[<u>-</u>	0.1		00 T	ignition
example 9								(Continued)

			Table 3		
	Battery	Initial	Reduction rate	High rate	High rate cycle
	temperature	discharge	of discharge	discharge	characteristics
	during nail	capacity	capacity after	characteristics	(%)
	sticking	[mAh]	300 cycles [%]	(%)	
	test $[^{\circ}C]$				
Comparative	117	1660	رد - 1	-41	-45
example 1	, T.T.))	H)	1	
Comparative	710	1660	131	l C C	ഗ
example 2	O T #	9	H		
Comparative	7	1740	 የ	-42	-43
example 3	0 1 1) - -)		15 (c)
Comparative	421	1740	-37	-57	1 58
example 4	т 1	1			
Comparative	7	1710	135	-41	-41
example 5)) - -			
Comparative	A 2 3	1710	132	- 59	-59
example 6) J) 			
Comparative	107	1690	- 35	-41	-42
example 7	7 7)))	, physican and a second	
Comparative	007	1700	98	ا ش	90 1
example 8	775) H			
Comparative		1680	-37	-57	159
example 9) 1 F)) 1	· •		

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As is apparent from Tables 1 to 3, the secondary battery for each of Examples 1 to 26 each comprising a positive electrode active material containing the composite oxide having the composition represented by chemical formula (1) referred to previously was free from rupture and ignition in the nail sticking test and was lower than the secondary batteries for Comparative Examples 1 to 9 in any of the discharge capacity reduction rate after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate in the high rate cycle in which the discharge rate was set at 5C.

Particularly, the secondary batteries for

Examples 4 to 15 each comprising the positive electrode active material containing Na, K or S in an amount falling within a range of between 1,000 ppm and 2,500 ppm were found to be lower than the secondary batteries for Examples 1 to 3 and 16 to 18 in the discharge capacity reduction rate in the high rate cycle. Further, the secondary battery for Example 23 comprising a positive electrode active material containing Ca in an amount not larger than 500 ppm was found to be superior to the secondary battery for Example 16 in the high rate discharge characteristics and the high rate cycle characteristics.

On the other hand, the secondary batteries for Comparative Examples 1, 3, 5 each comprising a positive

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electrode active material containing 500 ppm of Na, K or S and the secondary battery for Comparative Example 7 comprising a positive electrode active material containing 200 ppm of Ca were found to be larger than the secondary batteries for Examples 1 to 26 in any of the discharge capacity reduction rate after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate at the high rate cycle time in which the discharge rate was set at 5C.

Incidentally, a transmission electron microscopic observation was applied to the lithium-containing composite oxide used in the secondary battery for Example 1. It has been confirmed that, as shown in FIG. 4, Na metal was precipitated in triple points 33 (shaded regions) positioned in the boundaries among crystal grains 32.

Examples 27 to 49:

Prepared as starting materials were powders of

LiOH·H₂O, Ni(OH)₂, oxides, carbonates and nitrates of
the element Mel, NaOH, KOH, Ca(OH)₂, sodium sulfide
(Na₂S·9H₂O) as a sulfide compound, a sulfate
compound (NiSO₄·6H₂O), an oxide, a sulfide and
alkoxide of Si, and an oxide, a sulfide and an alkoxide
of Fe. These powdery compounds were selected to form
the composition shown in Tables 4 and 5, i.e.,
Li_{1.1}(Ni_{0.88}Mel_{0.02})(O_{1.9}X_{0.1}) + aA + bB, and mixed,

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followed by further mixing the composition in a Henschel mixer for 30 minutes so as to prepare a mixed powder. The mixed powder was put in an alumina sagger for firing. Firing was performed at 480° C for 10 hours while allowing oxygen gas to flow at a rate of 5 liter/min, followed by further firing the mixed powder at 700° C for 20 hours with an oxygen gas flow at a rate of 5 liter/min so as to obtain a positive electrode active material.

A cylindrical lithium ion secondary battery was prepared as in Example 1, except that the positive electrode active material thus prepared was used.

For the secondary battery prepared in each of Examples 27 to 49, measured were the occurrence or nonoccurrence of rupture and ignition by a nail sticking test, the battery temperature by the nail sticking test, the discharge capacity after the first cycle (initial discharge capacity), the reduction rate of the discharge capacity after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate at the high rate cycle in which the discharge rate was set at 5C as in Example 1. Tables 4 and 5 show the results.

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Table 4

(ppm) (ppm) test 3000 Fe 400 No rupture and n 3000 Fe 300 No rupture and n 2500 Fe 100 No rupture and n 2500 Fe 100 No rupture and n 2000 Fe 300 No rupture and n 2000 Fe 300 No rupture and n 2000 Si 400 No rupture and n 1000 Fe 300 No rupture and n 600 Si 100 No rupture and n 600 Si 100 No rupture and n 600 Si 200 No rupture and n 600 Fe 500 No rupture and n 600 Fe 500 No rupture and n 600 Si 200 No rupture and n 600 Si 200 No rupture and n 600 Si 200 No rupture and n 600 Si 300 No rupture and n			,			7.0	5	(а	ع	Result of nail sticking
Tatio False Fals		Li	Me	Motar	натодеп	ratio		(maa)	1	(wdd)	
X X X X X X X X X X		ratio		Y		2 2		4		4 1	
27 1.10 B 0.02 F 0.1 Na 3000 Fe 400 No rupture and no ignition 28 1.10 Mg 0.02 F 0.1 K 3000 Fe 300 No rupture and no ignition 29 1.10 M 0.02 F 0.1 N 2500 Fe 300 No rupture and no ignition 31 1.10 V 0.02 F 0.1 N 2500 Fe 100 No rupture and no ignition 32 1.10 V 0.02 F 0.1 N 2500 Fe 100 No rupture and no ignition 33 1.10 T 0.02 F 0.1 N 2000 Fe 50 No rupture and no ignition 34 1.10 T 0.02 F 0.1 N 1000 Fe 50 No rupture and no ignition 35 1.10 T 0.02 T 0.1 N 1000<		×									
28 1.10 Mg 0.02 F 0.1 K 3000 Fe 300 No rupture and no ignition 29 1.10 A1 0.02 F 0.1 S 3000 Fe 300 No rupture and no ignition 31 1.10 V 0.02 F 0.1 K 2500 Si 200 No rupture and no ignition 32 1.10 C 0.02 F 0.1 K 2500 Si 300 No rupture and no ignition 33 1.10 T 0.02 F 0.1 N 2000 Fe 100 No rupture and no ignition 34 1.10 T 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 35 1.10 T 0.02 F 0.1 N 2000 Fe 50 No rupture and no ignition 36 1.10 T 0.02 T 0.1 N 1000<	e 2	٠ .	В	•	ĹŦ.į	0.1	Na	3000	ъ	400	rupture and no
29 1.10 A1 0.02 F 0.1 S 3000 Fe 300 No rupture and no ignition 30 1.10 V 0.02 F 0.1 Na 2500 Si 200 No rupture and no ignition 31 1.10 C 0.02 F 0.1 K 2500 Si 300 No rupture and no ignition 32 1.10 Mn 0.02 F 0.1 K 2500 Si 300 No rupture and no ignition 33 1.10 Mn 0.02 F 0.1 K 2000 Si 400 No rupture and no ignition 34 1.10 C 0.02 F 0.1 K 2000 Si 400 No rupture and no ignition 35 1.10 C 0.02 F 0.1 K 1000 Fe 300 No rupture and no ignition 36 1.10 C 0.02 C 0.1 K 1	e 2	•	Mg	0.02	Ē	0.1	X	3000	Si	50	rupture and no
30 1.10 V 0.02 F 0.1 Na 2500 Si 200 No rupture and no ignition 31 1.10 Cr 0.02 F 0.1 K 2500 Si 300 No rupture and no ignition 32 1.10 Ti 0.02 F 0.1 K 2500 Fe 100 No rupture and no ignition 33 1.10 Ti 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 34 1.10 Cu 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 35 1.10 Cu Ci 0.1 K 1000 Fe 300 No rupture and no ignition 35 1.10 X 0.02 Cl 0.1 K 1000 Fe 300 No rupture and no ignition 34 1.10 X 0.02 Cl 0.1 K 600 <td< td=""><td>0</td><td></td><td>Al</td><td>0.02</td><td>ĹΉ</td><td>0.1</td><td>S</td><td>3000</td><td>ъ</td><td>300</td><td>rupture and no</td></td<>	0		Al	0.02	ĹΉ	0.1	S	3000	ъ	300	rupture and no
31 1.10 Cr 0.02 F 0.1 K 2500 Fe 100 No rupture and no ignition 32 1.10 Mn 0.02 F 0.1 S 2500 Si 300 No rupture and no ignition 33 1.10 T 0.02 F 0.1 K 2000 Fe 100 No rupture and no ignition 34 1.10 C 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 35 1.10 C 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 35 1.10 C 0.02 C1 0.1 K 100 Fe 300 No rupture and no ignition 38 1.10 F 0.02 C1 0.1 K 600 Fe 200 No rupture and no ignition 40 1.10 Mo 0.02 Br 0.1 K 600	ı m	• •	>	0.02	됴	0.1	Na	2500	Si	200	rupture and no
32 1.10 Mn 0.02 F 0.1 S 2500 Si 300 No rupture and no ignition 31 1.10 Ti 0.02 F 0.1 Na 2000 Fe 100 No rupture and no ignition 32 1.10 Cu 0.02 Cl 0.01 K 2000 Fe 350 No rupture and no ignition 32 1.10 Cu 0.02 Cl 0.01 K 1000 Fe 350 No rupture and no ignition 32 1.10 Cu 0.02 Cl 0.1 K 1000 Fe 350 No rupture and no ignition 32 1.10 Zn 0.02 Cl 0.1 K 1000 Fe 300 No rupture and no ignition 33 1.10 Zn 0.02 Cl 0.1 K 1000 Fe 300 No rupture and no ignition 33 1.10 Zn 0.02 Cl 0.1 K 600 Fe 300 No rupture and no ignition 34 1.10 No 0.02 Br 0.1 Na 1000 Fe 100 No rupture and no ignition 35 1.10 Ro 0.02 Br 0.1 Na 1000 Fe 100 No rupture and no ignition 36 1.10 Ro 0.02 Br 0.1 Na 1000 Fe 100 No rupture and no ignition 37 1.10 Ro 0.02 Br 0.1 Na 1000 Fe 100 No rupture and no ignition 38 1.10 Ro 0.02 Br 0.1 Ro 1000 Si 300 No rupture and no ignition 39 1.10 Ro 0.02 Br 0.1 Ro 1000 Si 300 No rupture and no ignition 30 1.10 Ro 0.02 Br 0.1 Ro 1000 Si 300 No rupture and no ignition 30 1.10 Ro 0.02 Br 0.1 Ro 1000 Si 300 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.02 Br 0.1 Ro 200 Fe 50 No rupture and no ignition 30 1.10 Sn 0.10 Sn			Cr	0.02	Ē	•	X	2500	ъ	100	rupture and no
33 1.10 Ti 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 3.1 1.10 Co 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 3.2 1.10 Cu 0.02 F 0.1 Na 1000 Fe 350 No rupture and no ignition 3.2 1.10 Ca 0.02 F 0.1 Na 1000 Fe 350 No rupture and no ignition 3.3 1.10 Ca 0.02 Cl 0.1 K 1000 Fe 30 No rupture and no ignition 3.3 1.10 Tr 0.02 Cl 0.1 Na 600 Si 100 No rupture and no ignition 4.0 1.10 Nb 0.02 Cl 0.1 Na 600 Si 100 No rupture and no ignition 4.0 1.10 Nb 0.02 Cl 0.1 Na 600 Si 200 No rupture and no ignition 4.0 1.10 Nb 0.02 Cl 0.1 Na 1100 Fe 100 No rupture and no ignition 4.1 1.10 Nb 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no ignition 5.1 1.10 Ru 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no ignition 5.1 1.10 Ru 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no ignition 5.1 1.10 Ru 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no ignition 5.1 1.10 Ru 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no ignition 5.1 1.10 Ru 0.02 Br 0.1 K 2800 Fe 50 No rupture and no ignition 6.1 Na 1.10 Sn 0.02 Br 0.1 K 2800 Fe 50 No rupture and no ignition 6.1 Na 1.10 Ca 0.00 Br 0.1 K 2800 Fe 50 No rupture and no ignition 6.1 Na 1.10 Ca 0.00 Br 0.1 K 2800 Fe 50 No rupture and no ignition 6.1 Na 1.10 Ca 0.00 Re 0.1 Na 1.10 Na 1		٠, ٠	Mn	0.02	ĮŦı	0.1	လ	2500	Si	300	rupture and no
34 1.10 Co 0.02 F 0.1 K 2000 Fe 50 No rupture and no ignition 35 1.10 Cu 0.02 F 0.1 S 2000 Si 400 No rupture and no ignition 36 1.10 Cu 0.02 F 0.1 K 1000 Fe 350 No rupture and no ignition 38 1.10 Y 0.02 Cl 0.1 K 1000 Fe 300 No rupture and no ignition 39 1.10 Y 0.02 Cl 0.1 K 600 Fe 200 No rupture and no ignition 40 1.10 Mo 0.02 Br 0.1 K 600 Si 200 No rupture and no ignition 41 1.10 Mo 0.02 Br 0.1 No 100 No rupture and no ignition 42 1.10 Ru 0.02 Br 0.1 No 100 No rupture and n	ļ		I.T.	0.02	됴		Na	2000	Б	100	rupture and no
35 1.10 Cu 0.02 Cl 0.1 Na 1000 Fe 350 No rupture and no ignition 36 1.10 Zn 0.02 F 0.1 K 1000 Si 50 No rupture and no ignition 38 1.10 X 0.02 Cl 0.1 K 1000 Fe 300 No rupture and no ignition 39 1.10 X 0.02 Cl 0.1 K 600 Fe 300 No rupture and no ignition 40 1.10 No 0.02 Cl 0.1 K 600 Fe 200 No rupture and no ignition 41 1.10 No 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no ignition 42 1.10 Ru 0.02 Br 0.1 Na 1100 No rupture and no ignition 43 1.10 Ru 0.02 Br 0.1 Na 100 No rupture	1		දි	0.02	됴		X	2000	ъ	50	rupture and no
36 1.10 Zn 0.02 F 0.1 Na 1000 Fe 350 No rupture and no ignition 37 1.10 Y 0.02 C1 0.1 X 1000 Fe 300 No rupture and no ignition 39 1.10 Y 0.02 C1 0.1 Na 600 Fe 200 No rupture and no ignition 40 1.10 Nb 0.02 C1 0.1 K 600 Fe 200 No rupture and no ignition 41 1.10 Mo 0.02 Br 0.1 K 600 Si 200 No rupture and no ignition 42 1.10 Mo 0.02 Br 0.1 Na 1100 No rupture and no ignition 42 1.10 Ru 0.02 Br 0.1 S 1000 Si 300 No rupture and no ignition 43 1.10 Ru 0.02 Br 0.1 K 200 No rupture a) (٠ ١	Cr	0.02	Cl	0.1	ល	2000	Si	400	rupture and no
37 1.10 Ga 0.02 C1 0.1 K 1000 Fe 300 No rupture and no ignition 38 1.10 Y 0.02 C1 0.1 Na 600 Si 100 No rupture and no ignition 40 1.10 Xr 0.02 C1 0.1 K 600 Fe 200 No rupture and no ignition 41 1.10 Mo 0.02 Br 0.1 S 600 Si 200 No rupture and no ignition 42 1.10 Mo 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no ignition 42 1.10 Ru 0.02 Br 0.1 S 1000 Si No rupture and no ignition 43 1.10 Ru 0.02 Br 0.1 K 280 No rupture and no ignition 44 1.10 Sn 0.02 Br 0.1 K 280 No rupture and no ignition) (١ ١	7,0	0.02	[조니	0.1	Na	1000	Fе	350	rupture and no ignition
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e 39 1.10 Zr 0.02 C1 0.1 K 600 Fe 200 No rupture and rupture and no rupture and rupture and no rup		' '	H	0.02	CJ	1 •	ß	1000	ъ	300	rupture and no
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e 41 1.10 Mo 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no e 42 1.10 Tc 0.02 Br 0.1 Na 1100 Fe 100 No rupture and no e 43 1.10 Ru 0.02 Br 0.1 S 1000 Si 300 No rupture and no e 44 1.10 Sn 0.02 Br 0.1 K 2800 Fe 50 No rupture and no c 44 1.10 Sn 0.02 Br 0.1 K 2800 Fe 50 No rupture and no c 50 No ruptu	4	1 .	qN	0.02	CJ	0.1	М	009	ъ	200	rupture and no
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e 43 1.10 Ru 0.02 Br 0.1 S 1000 Si 300 No rupture and no ca 44 1.10 Sn 0.02 Br 0.1 K 2800 Fe 50 No rupture and no Ca 200 Fe	e 4	•	TC		Br	0.1	Na	1100	FT O	100	rupture and no
e 43 1.10 Ru 0.02 Br 0.1 S 1000 Si 300 No rupture and no e 43 1.10 Sn 0.02 Br 0.1 K 2800 Fe 50 No rupture and no Ca 200 Ro Ca 200 Ro Ro Rupture and no Ca 200 Ro							Ca	200			
e 44 1.10 Sn 0.02 Br 0.1 K 2800 Fe 50 No rupture and no Ca 200	e 4	1	Ru	1 •	Br	0.1	ß	1000	Si	300	rupture and no
e 44 1.10 Sn 0.02 Br 0.1 K 2800 Fe 50 No rupture and no Ca 200		1					Ca	200			
Ca	e 4	1	Sn	•	Br	0.1	X	2800	ъ	50	rupture and no
							Ω O	200			

(Continued)

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			Table 4		
	Battery	Initial	Reduction rate	High rate	High rate cycle
	tu	discharge	of discharge	discharge	characteristics
	during nail	capacity	capacity after	characteristics	(%)
	sticking test	[mAh]	300 cycles [%]	0/0	
	ر م	1680	8 1	-13	-22
- 1	20 70 70	1710	-15	-16	-21
- 1	0) (0	1 7 1	-22
Example 29	96	1770	0 1	7	,
Example 30	96	1730	8	-13	ار ا
	0 5	1700	-17	-15	-14
- 1	5) C	1710	ი 1	-13	-13
Examble 32	20	> + · +		71-	-14
Example 33	96	1700	01-	. t	1 7
Example 34	96	1690	-13	-16	271
1	10	1690	-13	-14	-13
Example 33) H	L	6.1-	1,3
Example 36	97	1700	-T2	071) ,
Example 37	96	1700	-10	-17	-14
ì	86	1700	-16	-13	-13
Į.	97	1770	-11	-16	-20
		1750	110	-14	-22
Example 40	200	7	1	1 4	
Example 41	96	1680	6 1	-13	1
Fxample 42	95	1690	o -	6 1	5
	95	1700	-17	8	6 -
		1700	-12	ი I	-17
Examble 44		4			

(ppm) (ppm) 1000 1000 1000 200 Si 100 1000 Fe 200 1000 Fe 200 1000 Si 60 1500 Fe 200 1500 Fe 200 1500 Fe 200			1	Tab	' - '	e		α	ع,	Result of nail sticking
(ppm) (ppm) Cest Na 1500 Si 100 No rupture and no ignition K 600 Si 50 No rupture and no ignition S 1000 Fe 50 No rupture and no ignition S 1000 Fe 200 No rupture and no ignition Na 1000 Si 70 No rupture and no ignition K 1200 Si 60 No rupture and no ignition K 1200 Si 60 No rupture and no ignition K 1200 Si 200 No rupture and no ignition Na 2000 Si 200 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition		Me1	Molar	Halogen	Molar	₹	ช .	<u> </u>	(m c, c,	3 4 5 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6
Y X No rupture and no ignition S S S S S S S S S	Ħ		ratio		ratio		(mdd)		(mdd)	ר מ נ
Ta 0.02 I 0.1 Na 1500 Si 100 No rupture and no ignition Mg 0.02 F 0.1 K 600 Si 50 No rupture and no ignition Ti 0.02 F 0.1 S 1000 Fe 200 No rupture and no ignition Nb 0.02 C1 S 1500 Fe 200 No rupture and no ignition Nb 0.02 Br 0.1 S 1500 Fe 200 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 60 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 200 No rupture and no ignition	0		λ		Z					
Ta 0.02 I 0.1 Na 1500 S1 100 No rupture and no ignition Ca 100 Fe 50 No rupture and no ignition Ca 200 Fe 50 No rupture and no ignition Na 1000 Fe 200 No rupture and no ignition Nb 0.02 C1 0.1 S 1500 Fe 200 No rupture and no ignition K 1200 Si 60 No rupture and no ignition K 1200 Si 200 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 S 100	ļ						i	- ,	6	Me winting and no ignition
Mg 0.02 F 0.1 K 600 Si 50 No rupture and no ignition Ti 0.02 F 0.1 S 1000 Fe 50 No rupture and no ignition Nb 0.02 C1 0.1 S 1500 Fe 200 No rupture and no ignition K 1200 Si 60 No rupture and no ignition K 1200 Si 60 No rupture and no ignition K 1200 Si 60 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 S		ΕĦ	0.02	Н	0.1	Na	1500	S1	OOT	NO IUDIULE aila ilo railicae
Mg 0.02 F 0.1 K 600 Si 50 No rupture and no ignition Ti 0.02 F 0.1 S 1000 Fe 200 No rupture and no ignition Nb 0.02 C1 0.1 S 1500 Fe 200 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 60 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 200 No rupture and no ignition No 1000 Fe 100 Fe 100 Fe 100	,					വ	1000			
Mg 0.02 F 0.1 K 600 Si 50 No rupture and no ignition Ti 0.02 F 0.1 S 1000 Fe 200 No rupture and no ignition Nb 0.02 C1 0.1 S 1500 Fe 200 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 60 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 200 No rupture and no ignition S 1000 Fe 100 Fe 100						Ca	100			
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Ti 0.02 F 0.1 S 1000 Fe 200 No rupture and no ignition Nb 0.02 C1 S 1500 Fe 200 No rupture and no ignition K 1200 Si 60 No rupture and no ignition K 1200 Si 60 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 1000 Fe 1000 S 1000 Fe 1000 S 1000 Fe 1000 S 1000 Fe 1000 S 1000	>	617	3			C a	200	ъi О	50	
Nb 0.02 C1 0.1 S 1500 Fe 200 No rupture and no ignition K 1200 Si 60 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 200 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 100 No rupture and no ignition S 1000 Fe 1000 Fe 1000 Fe 1000 S 1000 Fe 1000 S 1000	0	- L	0 02	[E	0.1	ß	1000	ъ	200	No rupture and no ignition
Nb 0.02 Cl 0.1 S 1500 Fe 200 No rupture and no ignition Mo 0.02 Br 0.1 Na 2000 Si 60 No rupture and no ignition S 1000 Fe 100 Fe 100	>	1	9			Na	1000	Si	70	
Mo 0.02 Br 0.1 Na 2000 Si 60 500 5	(ZIZ ZIZ		CJ	0.1	ß	1500	Fе	200	No rupture and no ignition
Mo 0.02 Br 0.1 Na 2000 Si 200 S 1000 Fe 100	>	2	_			X	1200	Si	09	
S 1000 Fe		Ž	-	_	0.1	Na	2000	Si	200	No rupture and no ignition
	>	OF T	+	-		ß	1000	Eri (I)	100	

	High rate cycle characteristics (%)	C	ח	-16	0.7	711	-13	1		
	High rate discharge characteristics (%)		ω	-13	OT-	-13	-14	0.7	CT.	
Table 5	rate rge after s [%]		თ 		20 1	ا ئ	αI		9-	
	Initial discharge capacity [mAh]		1730	7	1730	1760	000	08/T	1710	
	Battery temperature during nail sticking test	[0]	90	30	96	91	1)	06	91	
			L	Example 45	Example 46		EXample 4/	Example 48	Example 49	o - Amount

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As is apparent from Tables 4 and 5, the secondary battery for each of Examples 27 to 49 each comprising a positive electrode active material containing the composite oxide having the composition represented by chemical formula (2) referred to previously was free from rupture and ignition in the nail sticking test and was lower than the secondary batteries for Comparative Examples 1 to 8 in any of the discharge capacity reduction rate after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate in the high rate cycle in which the discharge rate was set at 5C. Further, the secondary batteries for Examples 27 to 49 were lower than the secondary batteries for Examples 1 to 26 in the battery temperature in the step of the nail sticking test.

Particularly, the secondary batteries for Examples 30 to 38 each comprising the positive electrode active material containing Na, K or S in an amount falling within a range of between 1,000 ppm and 2,500 ppm were found to be lower than the secondary batteries for Examples 27 to 29 and 39 to 41 in the discharge capacity reduction rate in the high rate cycle. Further, the secondary batteries for Examples 42 to 46 each comprising a positive electrode active material containing Ca in an amount not larger than 500 ppm were found to be superior to the secondary battery in which

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Ca was not added to the positive electrode active material in the high rate discharge characteristics and the high rate cycle characteristics.

Examples 50 to 69 and Comparative Examples 10 to 13:

Prepared as starting materials were powders of LiOH·H₂O, Ni(OH)₂, Co(OH)₂, oxides, carbonates and nitrates of the element Me2, NaOH, KOH, Ca(OH)2, sodium sulfide (Na₂S · 9H₂O) as a sulfide compound, and a sulfate compound (NiSO4 · 6H2O). These powdery compounds were selected to form the composition shown in Tables 6 and 7, i.e., $Li_{1.1}(Ni_{0.70}Co_{0.18}Me_{20.02})(O_{1.9}X_{0.1}) + aA$, and mixed, followed by further mixing the composition in a Henschel mixer for 30 minutes so as to prepare a mixed powder. The mixed powder was put in an alumina sagger for firing. Firing was performed at 480° C for 10 hours while allowing oxygen gas to flow at a rate of 5 liter/min, followed by further firing the mixed powder at 700° C for 20 hours with an oxygen gas flow at a rate of 5 liter/min so as to obtain a positive electrode active material.

A cylindrical lithium ion secondary battery was prepared as in Example 1, except that the positive electrode active material thus prepared was used.

For the secondary battery prepared in each of Examples 50 to 69 and Comparative Examples 10 to 13,

measured were the occurrence or nonoccurrence of rupture and ignition by a nail sticking test, the battery temperature by the nail sticking test, the discharge capacity after the first cycle (initial discharge capacity), the reduction rate of the discharge capacity after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate at the high rate cycle in which the discharge rate was set at 5C as in Example 1. Tables 6 and 7 show the results.

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					Table	e 6			-
	1,1	CO	Me2	Molar	Halogen	Molar	A	ರ	Result of mail sticking
	molar	molar		ratio		ratio		(wdd)	test
	ratio	ratio		ω		7			
	×	۸				1			1
Fyamnle 50	1.10	0.18	В	0.02	[Ŧ4	0.1	Na	3000	Tupture and no
ر ا د	٠ ا٠	0.18	Mg	0.02	[五	0.1	X	1500	rupture and no
- 1	1 .	0.18	Al	0.02	ĹΤΙ	0.1	ß	2500	rupture and no
i	•	0.18	Sc	0.02	[Ŧ1	0.1	X	1000	and mo
- 1	•	: -	M	0.02	[zu	0.1	ß	3000	and no
ט∫ י	•	1 -	5	0.02	CJ	0.1	Na	1500	No rupture and no ignition
Example 55			3	•			Ç	300	
					5	0	Na	1000	No rupture and no ignition
Example 56	1.10	0.18	Zn	0.02	3 5		4	3000	No rupture and no ignition
Fyample 57	1.10	0.18	Ga	0.02	CJ	7.0	4		
l l	7	0 1 8	X	0.02	Cl	0.1	လ	1700	alla
Example 58	-	; - -	E	0	R	0.1	Na Na	009	No rupture and no ignition
Example 59			ည -	0.0	-	0	ι Ω	1200	No rupture and no ignition
Example 60	1.10		+	0.02			V.	009	No rupture and no ignition
Example 61	1.10	0.18	Sn	0.02	\\ -\-		ט נ	1500	No rupture and no ignition
Eyample 62	1.10	0.18	Гa	0.02		7.0	2		
Į			_				Ca	100	
- 1	-			0 00	 	0.1	M	009	No rupture and no
9	-		-	-		0.1	X	1000	No rupture and no ignition
Example 64	4 1.10	0 0		-	-		Ca	200	
									(Continued)

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			Table 6		r
	Battery	Initial	Reduction rate	High rate	High rate cycle
	temperature	discharge	of discharge	discharge	characteristics
	during nail	capacity	capacity after	characteristics	96
	sticking test	[mAh]	300 cycles [%]	(%)	
	[,C]				
Example 50	98	1580	8 1	-13	77.
1	86	1560	-15	-16	115
- 1	90	1600	ი !	-13	-14
1	00	0 0	114	-15	-15
Example 53	87	OGCT	# 	7	-21
Example 54	98	1570	8	۲ <u>۱</u>	17
1 2 0 Cmcv	06	1560	-12	δ I	ו
1		1550	80	-14	-13
Example 50		9	τ	1 2	-22
Example 57	91	1560	-15		
Example 58	92	1610	6 1	7 1.	# C
Fxample 59	06	1550	-14	-13	07-
	92	1550	8	-13	211
		1580	-14	-14	-21
		1600		6 -	ω 1
Example 62	אַ	O O O O	1 7		120
Example 63	91	1570	-13	0	
Example 64	92	1560	-11	5	6 1
+					

Table 7

	-,	ť	(),	M() 10 %	Table Table	Moler	d	π	Result of nail sticking
	molar	molar	701.7	ratio	1100000	ratio	i i	(wdd)	
	ratio	ratio		מ		Ŋ			
	×	V							
Example 65	1.10	0.18	В	0.02	Ή	0.1	Na	2000	No rupture and no ignition
	1						ಬ	800	
							Ca	200	
F.x.amp]e 66	1.10	0.18	Mq	0.02	ĒΉ	0.1	Ж	1500	No rupture and no ignition
	•						Ca	100	
							Na	1500	
							S	1100	
Example 67	1.10	0.18	Al	0.02	<u>ਜਿ</u>	0.1	S	2500	No rupture and no ignition
1	1						Na	1500	
							Ca	100	
Example 68	1.10	0.18	Cu	0.02	CI	0.1	Ca	300	No rupture and no ignition
1	1						S	1800	
							K	1200	
Example 69	1.10	0.18	G H	0.02	Br	0.1	K	1000	No rupture and no ignition
	1	1					S	1700	
							Na	1500	
Comparative	1.10	0.18	В	0.02	DZı	0.1	none	l	Occurrence of rupture and ignition
1 1	1.10	0.18	CG	0.02	C]	0.1	none	1	Occurrence of rupture and ignition
1 11	1.10	0.18	Sn	0.02	Br	0.1	none	I	Occurrence of rupture and ignition
\perp	1.10	0.18	La	0.02	H	0.1	none	1	Occurrence of rupture and ignition
1		1							(Continued)

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						1			т-			Т"		
	High rate cycle	characteristics	(%)			8 1	6 1	6 -	6 -	8 -	09-	-61	-61	09-
	High rate	discharge	characteristics	(%)		ත 	8	ი 	80 1	80 1	- 58	1 59	-57	158
Table /	Reduction rate	of discharge	capacity after	300 cycles [%]		I V	9 -	l N	7 -	L -	-31	-30	-38	-35
	Initial	discharge	capacity	[mAh]		1570	1580	1630	1580	1570	1560	1540	1560	1580
	Battery	temperature	during nail	sticking test	[,C]	81	81	80	83	84	410	423	425	431
						Example 65	Example 66	Example 67	Example 68	Example 69	Comparative example 10	Comparative example 11	Comparative example 12	Comparative

As is apparent from Tables 6 and 7, the secondary battery for each of Examples 50 to 69 each comprising a positive electrode active material containing the composite oxide having the composition represented by chemical formula (3) referred to previously was free from rupture and ignition in the nail sticking test and was lower than the secondary batteries for Comparative Examples 1 to 13 in any of the discharge capacity reduction rate after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate in the high rate cycle in which the discharge rate was set at 5C.

Particularly, the secondary batteries for Examples 51 to 53, 56, 58 and 60 each comprising the positive electrode active material containing Na, K or S in an amount falling within a range of between 1,000 ppm and 2,500 ppm were found to be lower than the secondary batteries for Examples 50, 54, 57, 59, 61 and 63, in which the content of Na, K or S did not fall within the range of between 1,000 ppm and 2,500 ppm, in the discharge capacity reduction rate at the time of the high rate cycle in which the discharge rate was set at 5C. Further, the secondary battery for Example 64 comprising a positive electrode active material containing Ca in an amount not larger than 500 ppm was found to be superior to the secondary battery in which Ca was not added to the positive electrode active

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material in the high rate discharge characteristics and the high rate cycle characteristics.

Examples 70 to 89:

Prepared as starting materials were powders of LiOH · H₂O, Ni(OH)₂, Co(OH)₂, oxides, carbonates and nitrates of the element Me2, NaOH, KOH, Ca(OH)2, sodium sulfide (Na₂S·9H₂O) as a sulfide compound, a sulfate compound (NiSO $_4 \cdot 6H_2O$), an oxide, a sulfide and an alkoxide of Si, and an oxide, a sulfide and an alkoxide These powdery compounds were selected to form the composition shown in Tables 8 and 9, i.e., $\text{Li}_{1,1}(\text{Ni}_{0,70}\text{Co}_{0,18}\text{Me}_{20,02})(\text{O}_{1,9}\text{X}_{0,1}) + a\text{A} + b\text{B}$, and mixed, followed by further mixing the composition in a Henschel mixer for 30 minutes so as to prepare a mixed powder. The mixed powder was put in an alumina sagger for firing. Firing was performed at $480^{\circ}\mathrm{C}$ for 10 hours while allowing oxygen gas to flow at a rate of 5 liter/min, followed by further firing the mixed powder at 700° C for 20 hours with an oxygen gas flow at a rate of 5 liter/min so as to obtain a positive electrode active material.

A cylindrical lithium ion secondary battery was prepared as in Example 1, except that the positive electrode active material thus prepared was used.

For the secondary battery prepared in each of Examples 70 to 89, measured were the occurrence or nonoccurrence of rupture and ignition by a nail

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sticking test, the battery temperature by the nail sticking test, the discharge capacity after the first cycle (initial discharge capacity), the reduction rate of the discharge capacity after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate at the high rate cycle in which the discharge rate was set at 5C as in Example 1. Tables 8 and 9 show the results.

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C	2
E	-

						Table	ω ω				
	Li	පි	Me2	Molar	Halogen	Molar	A	ø	В	Д	Result of nail sticking
	molar	molar		ratio		ratio		(mdd)		(wdd)	test
	ratio	ratio		Ø		N					
	×	Λ									
Example 70	1.10	0.18	В	0.02	Ŀı	0.1	Na	3000	Fe	400	No rupture and no ignition
Example 71	1.10	0.18	Mg	0.02	Ĺч	0.1	X	1500	Si	50	No rupture and no ignition
Example 72	1.10	0.18	Al	0.02	[Ŧı	0.1	S	2500	ъe	300	No rupture and no ignition
Example 73	1.10	0.18	SC	0.02	ĮΞι	0.1	K	009	Si	100	No rupture and no ignition
Example 74	1.10	0.18	Mn	0.02	ĹΉ	0.1	ಬ	2200	ъ	200	No rupture and no ignition
Example 75	1.10	0.18	Cu	0.02	Cl	0.1	K	1000	Si	200	No rupture and no ignition
							Ca	300			
Example 76	1.10	0.18	Zn	0.02	Cl	0.1	Na	1000	ъ	100	No rupture and no ignition
Example 77	1.10	0.18	Ga	0.02	C1	0.1	X	3000	Si	300	No rupture and no ignition 6
Example 78	1.10	0.18	X	0.02	C1	0.1	വ	3000	F.	50	No rupture and no ignition
Example 79	1.10	0.18	Tc	0.02	Br	0.1	Na	2500	Si	400	No rupture and no ignition
Example 80	1.10	0.18	Ru	0.02	Br	0.1	ß	1200	ъ	400	No rupture and no ignition
Example 81	1.10	0.18	Sn	0.02	Br	0.1	×	2100	Si	50	No rupture and no ignition
Example 82	1.10	0.18	La	0.02	Н	0.1	Na	1500	E O	300	No rupture and no ignition
							Ç	200			
Example 83	1.10	0.18	Re	0.02	I	0.1	S	009	Si	100	No rupture and no ignition
Example 84	1.10	0.18	Pb	0.02	Н	0.1	ß	1500	ъ	200	No rupture and no ignition
							Ça	150			
Example 85	1.10	0.18	Bi	0.02	Н	0.1	Na	009	Si	200	No rupture and no ignition

(Continued)

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			Table 8		
	battery	Initial	Reduction rate	High rate	High rate cycle
	temperature	discharge	of discharge	discharge	characteristics
	[2]	capacity	capacity after	characteristics	(%)
		[mAh]	300 cycles [%]	(%)	The state of the s
Example 70	82	1600	8 -	-13	-21
Example 71	82	1570	-15	-16	-15
Example 72	83	1620	6 1	-14	-14
Example 73	83	1580	-14	-16	-21
Example 74	82	1590	& 	-13	-14
Example 75	98	1570	-12	6	6 -
Example 76	98	1570	ω	-13	-13
Example 77	87	1580	-15	-15	-22
Example 78	86	1640	б 1	-13	-20
Example 79	98	1580	6 1	-13	-13
Example 80	8.7	1570	ω	-14	-13
Example 81	87	1600	-13	-15	-15
Example 82	8.7	1620	-11	6 -	6 -
Example 83	87	1590	-10	-14	-21
Example 84	86	1570	-10	6 -	6 -
Example 85	86	1570	8 I	-14	-21

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Table 9

										-	<u>-</u>	97			ıre		ıre		ıre		ıre	_
Result of nail	sticking test			No rupture and no ignition			No rupture and no ignition			No rupture and no ignition			No rupture and no ignition		Occurrence of rupture	and ignition	Occurrence of rupture	and ignition	Occurrence of rupture	and ignition	Occurrence of rupture	
b R	s (mdd)			400 li	09	1	50 N	400	I	300	70	ı	50 N	400	١		Ō			ਰ		
В				ъ	Si	ì	Si	Fe	_	Fi O	Si		Si	Fe	000	יוסוום	, ,	11011	(ווסווע	\$ \$	1
ď	(mdd)			2000	1500	250	1500	1200	150	2500	1200	1000	1000	1200	1		1					ı
A				Na	ಬ	Ca	Ж	S	Ca	ß	Na	K	Ж	ಬ	o to	ווסווב	0	11011	Ş	ווסוות	(4
Molar	ratio	Z		0.1			0.1			0.1			0.1		1	0.1	,	0. I	,	T•0	,	
Halogen				伍			뇬			ĹŦI			Br		Ŀ	T.	ົເ	J	Ę.	DI	ŀ	_
Molar	ratio	ഗ		0.02			0.02			0.02	•		0.02		CO	0.02	CO	70.0		0.02	C C	
Me2				В			Mg			Al			Sn		р	Δ	ć	ت ک	;	211	F	α
3	molar	ratio	٥	0.18			0.18			0.18			0.18		0 7	01.0	0	0.1.0	0	0.10		×
Li	molar	ratio	×	1.10	The state of the s		1.10	investigation of the state of t		1.10	The second secon		1.10		-	O T • T	,	OT•T	,	ОТ•Т	7	=
				Example 86			Example 87			Example 88			Example 89	Action of the second of the se	Comparative	example 10	Comparative	example 11	Comparative	example 12	Comparative	

(Continued)

			Table 9		er (Fernande)
	battery	Initial	Reduction rate	High rate	High rate cycle
	temperature	discharge	of discharge	discharge	characteristics
	[°C]	capacity	capacity after	characteristics	(%)
		[mAh]	300 cycles [%]	(%)	
Example 86	80	1590	- 5	8	6 1
Example 87	76	1600	- 6	8 -	8 1
Example 88	78	1650	- 5	-13	-13
Example 89	81	1620	L -	-14	-14
Comparative	410	1560	131	1 10 10	09-
example 10)) 	1		
Comparative	861	1540	-30	ا 55	-61
example 11	C 7 F))		
Comparative	/ የ	ر م	α κ 1	-57	1 61
example 12	O N T)) H)		
Comparative	131	ر م	ا س ت	ا تن ش	09-
example 13	ヿヿ)) 1))	and the state of t

As is apparent from Tables 8 and 9, the secondary battery for each of Examples 70 to 89 each comprising a positive electrode active material containing the composite oxide having the composition represented by chemical formula (4) referred to previously was free from rupture and ignition in the nail sticking test and was lower than the secondary batteries for Comparative Examples 1 to 13 in any of the discharge capacity reduction rate after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate in the high rate cycle in which the discharge rate was set at 5C.

Particularly, the secondary batteries for
Examples 71, 72, 74, 76 and 79 to 81 each comprising
the positive electrode active material containing Na, K
or S in an amount falling within a range of between
1,000 ppm and 2,500 ppm were found to be lower than the
secondary batteries for Examples 70, 73, 77, 78, 83 and
85, in which the content of Na, K or S did not fall
within the range of between 1,000 ppm and 2,500 ppm, in
the discharge capacity reduction rate at the time
of the high rate cycle in which the discharge rate was
set at 5C. Further, the secondary batteries for
Examples 75, 82, 84, 86 and 87 each comprising
a positive electrode active material containing Ca in
an amount not larger than 500 ppm were found to be
capable of markedly improving the high rate discharge

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characteristics and the high rate cycle characteristics.

Examples 90 to 105:

Prepared as starting materials were powders of LiOH · H₂O, Ni(OH)₂, Co(OH)₂, oxides, carbonates and nitrates of the element Me3, NaOH, KOH, Ca(OH)2, sodium sulfide (Na₂S·9H₂O) as a sulfide compound, and a sulfate compound (NiSO4 · 6H2O). These powdery compounds were selected to form the composition shown in Tables 10 and 11, i.e., $Li_{1.1}(Ni_{0.70}Co_{0.18}Me_{30.02})(O_{1.9}X_{0.1}) + aA$, and mixed, followed by further mixing the composition in a Henschel mixer for 30 minutes so as to prepare a mixed The mixed powder was put in an alumina sagger for firing. Firing was performed at 480° C for 10 hours while allowing oxygen gas to flow at a rate of 5 liter/min, followed by further firing the mixed powder at 700° C for 20 hours with an oxygen gas flow at a rate of 5 liter/min so as to obtain a positive electrode active material.

A cylindrical lithium ion secondary battery was prepared as in Example 1, except that the positive electrode active material thus prepared was used.

For the secondary battery prepared in each of Examples 90 to 105, measured were the occurrence or nonoccurrence of rupture and ignition by a nail sticking test, the battery temperature by the nail

sticking test, the discharge capacity after the first cycle (initial discharge capacity), the reduction rate of the discharge capacity after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate at the high rate cycle in which the discharge rate was set at 5C as in Example 1. Tables 10 and 11 show the results.

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					Table	e 10			
	Li	တ္	Me3	Molar	Halogen	Molar	A	ď	Result of nail sticking
	molar	molar		ratio		ratio		(mdd)	test
	ratio	ratio		Ţ		Z		-	
	×	Δ							
Example 90	1.10	0.18	Τi	0.02	ĹΉ	0.1	ß	1500	No rupture and no ignition
1							Сa	200	
Example 91	1.10	0.18	>	0.02	ᄄ	0.1	Na	1500	No rupture and no ignition
	1.10	0.18	Cr	0.02	ĒŦ	0.1	Х	1000	No rupture and no ignition
	1.10	0.18	Zr	0.02	C1	0.1	ß	2500	No rupture and no ignition
1	1.10	0.18	QN QN	0.02	CJ	0.1	Na	1500	No rupture and no ignition
						ı	Ca	200	
T 1 C 2 C.	1 10	0.18	Mo	0.02	Br	0.1	Na	2500	No rupture and no ignition
i	1.10	0.18	H	0.02	Br	0.1	Na	009	No rupture and no ignition
- 1	1.10	0.18	EH B	0.02	Н	0.1	저	009	No rupture and no ignition
	1.10	0.18	M	0.02	Н	0.1	ဃ	009	No rupture and no ignition
1	1.10	0.18	Ti	0.02	Ē	0.1	Ca	200	No rupture and no ignition
1							Na	1200	
							ഗ	1500	

(Continued)

Table 10

	Battery	Initial	Reduction rate	High rate	High rate cycle
	temperature	discharge	of discharge	discharge	characteristics
	during nail	capacity	capacity after	characteristics	(%)
	sticking	[mAh]	300 cycles [%]	(%)	
	test [%]				
Example 90	79	1720	-11	6 -	6 -
	79	1700	∞ 1	-13	-14
	79	1680	116	-15	-19
1	82	1680	6	-14	-13
	83	1750	-12	6 -	ω
		1680	8	-13	-14
1		1680	8 1	-13	-20
1		1750	-17	-16	-20
Example 98	83	1710	თ 1	-14	-20
Example 99	68	1720	l S	6 -	ر 0

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					Table	11			
	Li	දි	Me3	Molar	Halogen	Molar	A	rd	Result of nail sticking
	molar	molar		ratio		ratio		(mdd)	test
	ratio	ratio		4		Z			
	×	Λ							
Example 100	1.10	0.18	Λ	0.02	ſΞŧ	0.1	Na	1500	No rupture and no ignition
1							Ca	100	
							S	1500	
Example 101	1.10	0.18	Cr	0.02	H	0.1	又	1000	No rupture and no ignition
							Na	2000	
							ß	1500	
Example 102	1.10	0.18	Zr	0.02	C1	0.1	S	3000	No rupture and no ignition
							Na	1000	
Example 103	1.10	0.18	qN	0.02	C1	0.1	Сa	200	No rupture and no ignition
4							內	1500	
							ß	1000	
Example 104	1.10	0.18	Mo	0.02	Br	0.1	Na	3000	No rupture and no ignition
							X	1100	
							ß	1000	
Example 105	1.10	0.18	Та	0.02	Н	0.1	А	1000	No rupture and no ignition
							Ca	150	
							Na	1500	
									(polia: + #05)

(Continued)

			Table 11	7,000		
	Battery	Initial	Reduction rate	High rate	High rate cycle	
	temperature	discharge	of discharge	discharge	characteristics	
	during nail	capacity	capacity after	characteristics	(%)	
	sticking	[mAh]	300 cycles [%]	(%)		
	test					
	[°C]			and the state of t		
Example 100	7.0	1700	-5	ර 	8 1	
Example 101	7.0	1680	-5	-13	-13	
Example 102	72	1680	-5	-13	-22	
Example 103	73	1750	-8	8	თ 	
Example 104	73	1680	-5	-14	-21	
Example 105	92	1750	-7	ර I	6 -	

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As is apparent from Tables 10 and 11, the secondary battery for each of Examples 90 to 105 each comprising a positive electrode active material containing the composite oxide having the composition represented by chemical formula (5) referred to previously was free from rupture and ignition in the nail sticking test and was lower than the secondary batteries for Comparative Examples 1 to 13 in any of the discharge capacity reduction rate after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate in the high rate cycle in which the discharge rate was set at 5C.

Particularly, the secondary batteries for
Examples 91 to 93 and 95 each comprising the positive
electrode active material containing Na, K or S in
an amount falling within a range of between 1,000 ppm
and 2,500 ppm were found to be lower than the secondary
batteries for Examples 96 to 98, in which the content
of Na, K or S did not fall within the range of between
1,000 ppm and 2,500 ppm, in the discharge capacity
reduction rate at the time of the high rate cycle in
which the discharge rate was set at 5C. Further, the
secondary batteries for Examples 90, 94, 99, 100, 103,
and 105 each comprising a positive electrode active
material containing Ca in an amount not larger than
500 ppm was found to be capable of suppressing each of

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the capacity reduction rate at the time of the high rate discharge and the discharge capacity reduction rate at the time of the high rate cycle to a level not higher than 10%.

Examples 106 to 120:

Prepared as starting materials were powders of LiOH· H_2O , Ni(OH)₂, Co(OH)₂, oxides, carbonates and nitrates of the element Me3, NaOH, KOH, Ca(OH)2, sodium sulfide (Na₂S · 9H₂O) as a sulfide compound, a sulfate compound (NiSO $_4$ \cdot 6H $_2$ O), an oxide, a sulfide and an alkoxide of Si, and an oxide, a sulfide and an alkoxide of Fe. These powdery compounds were selected to form the composition shown in Tables 12 and 13, i.e., $\text{Li}_{1.1}(\text{Ni}_{0.70}\text{Co}_{0.18}\text{Me}_{30.02})(\text{O}_{1.9}\text{X}_{0.1}) + \text{aA} + \text{bB}$, and mixed, followed by further mixing the composition in a Henschel mixer for 30 minutes so as to prepare a mixed powder. The mixed powder was put in an alumina sagger for firing. Firing was performed at $480^{\circ}\mathrm{C}$ for 10 hours while allowing oxygen gas to flow at a rate of 5 liter/min, followed by further firing the mixed powder at $700^{\circ}\mathrm{C}$ for 20 hours with an oxygen gas flow at a rate of 5 liter/min so as to obtain a positive electrode active material.

A cylindrical lithium ion secondary battery was prepared as in Example 1, except that the positive electrode active material thus prepared was used.

For the secondary battery prepared in each of

Examples 106 to 120, measured were the occurrence or nonoccurrence of rupture and ignition by a nail sticking test, the battery temperature by the nail sticking test, the discharge capacity after the first cycle (initial discharge capacity), the reduction rate of the discharge capacity after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate at the high rate cycle in which the discharge rate was set at 5C as in Example 1. Tables 12 and 13 show the results.

				tion		tion	ignition	tion		10		ignition '	ignition	ignition	tion			16.
Result of nail sticking	test			No rupture and no ignition		No rupture and no ignition	No rupture and no igni	No rupture and no ignition	No rupture and no ignition		No rupture and no ignition	No rupture and no igni	No rupture and no igni	No rupture and no igni	No rupture and no ignition			
q	(wdd)			300		50	200	100	100		200	50	300	150	300	100		
М				ы		Si	Fe	Si	ъ	-	Si	면 e	Si	ь	Б Б	Si		
ď	(wdd)			2000	200	1500	2500	600	1500	200	009	3000	1500	1200	400	1000	2000	
A				Na	Ca	Na	ß	S	S	Ç	Na	Na	X	ß	Ca	Na	က	
Molar	ratio	Z		0.1		0.1	0.1	0.1	0.1		0.1	0.1	0.1	0.1	0.1			
Haloge	n			[zi		Íτι	됴	당	CI		Br	Br	Н	Н	Ē			
Molar	ratio	ىد		0.02		0.02	0.02	0.02	0.02		0.02		0.02	0.02	0.02			
Me3				Ti		Λ	Cr	Zr	QN QN		Σ	P H	e	M	Ę	 		
8	lar	ratio	Δ	0.18		0.18	0.18	0.18	0.18		7	0 0	0 18	2 0	7 0)		
ן ינו	lar	ratio	×	1.10		1.10	1.10	1 10	1.10		1 10	1 -	1 0	1 10	7	- - - -		
				Example 106	4	Fxample 107					() () () () () () () () () ()		1	1		ביא שדכוווסדם		

(Continued)

			Table 12		
	ha+terv	Initial	Reduction rate	High rate	High rate cycle
	temperature		of discharge	discharge	characteristics
		capacity	capacity after	characteristics	(o/o
	1	[mAh]	300 cycles [8]	(%)	
901 - 1	00	1740	-12	6	ත 1
Example 100		1730	1	-13	-14
Example 10/) () () () () () () () () () () () () ()	1	-14	-14
Example 108	69	00/T		1	
Tvamp10 109	73	1680	6 1	-14	07-
Cot ot dimount		1750	1	& 1	6 -
Example 110	7/	00/T	4	7	-21
Fxample 111	73	1680	8 1	-13	T7_
110	73	1770	& 1	-14	-22
FXampre 117		0 0	1 7	1 5	-14
Example 113	73	T / 30	7		7
Example 114	73	1680	6	-13	7 1
	61	1740	ا 5	8	o .
Evanipre TT					

(Continued)		ŀ			5						
	100	Si	1000	ೱ							
ignition								:			ozi ardiibxa
No rupture and no	150	Fe Fe	2000	လ	0.1	Н	0.02	ß	0.18	1 10	120 120
	150	Si	1000	ᅜ							
ignition											
No rupture and no	20	급 의	1500	Na	0.1	Br	0.02	Hf	0.18	1 10	Tvample 119
	80	Si	1000	Na							
ignition								<u> </u>			orr ardinorr
No rupture and no	100	т Э	200	Ca	0.1	CI	0.02	Q	0.18	1 10	Fyample 118
			2000	ಬ							
	70	Si	150	Ca							
ignition					_			:) - -		/ TT ATGINDYT
No rupture and no	200	н Ө	1000	М	0.1	[<u></u>	0.02	Cr	0.18	1 10	Tramale 117
			200	Ca							
	150	ъ	1000	S							
no Inplume and including ignition	00	S1	1500	Na	0.1	Ŀч	0.02	\triangleright	0.18	1.10	Example 116
7. Carritor			1						Δ	×	
					N		ب		ratio	ratio	
sticking test	(wdd)	-	(wdd)	-	ratio		ratio		molar	molar	
Result of half	.Ω	<u>—</u>	Ø	A	Molar	Halogen	Molar	Me3	ည	Li	
		ľ		ļ	Table 13	Па					
					ָר רַרָּ	E					

γ,	`
τ-	+
7	4
T o L	לק

	battery	Initial	Reduction rate High rate	High rate	High rate cycle
	temperature	discharge	temperature discharge of discharge	discharge	characteristics
	[2]	capacity	ы	characteristics	(%)
		[mAh]	300 cycles [%]	(%)	
Example 116	61	1730	ا 5	6 1	6 -
Example 117	61	1700	8	∞ I	6
Example 118	6.5	1770	9 -	6 -	<u>ი</u>
Example 119	99	1680	8 1	-13	-14
	99	1710	110	114	ا ا

As is apparent from Tables 12 and 13, the secondary battery for each of Examples 106 to 120 each comprising a positive electrode active material containing the composite oxide having the composition represented by chemical formula (6) referred to previously was free from rupture and ignition in the nail sticking test and was lower than the secondary batteries for Comparative Examples 1 to 13 in any of the discharge capacity reduction rate after 300 cycles, the discharge capacity reduction rate at the discharge at 5C, and the discharge capacity reduction rate in the high rate cycle in which the discharge rate was set at 5C.

Particularly, the secondary batteries for
Examples 107, 108, 113 and 114 each comprising the
positive electrode active material containing Na, K or
S in an amount falling within a range of between
1,000 ppm and 2,500 ppm were found to be lower than
the secondary batteries for Examples 109, 111 and 112,
in which the content of Na, K or S did not fall within
the range of between 1,000 ppm and 2,500 ppm, in
the discharge capacity reduction rate at the time of
the high rate cycle in which the discharge rate was
set at 5C. Further, the secondary batteries for
Examples 106, 110, 115 and 116 to 118 each comprising
a positive electrode active material containing Ca in
an amount not larger than 500 ppm were found to be

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capable of suppressing each of the capacity reduction rate at the time of the high rate discharge and the discharge capacity reduction rate at the time of the high rate cycle to a level not higher than 10%.

Example 121:

<Preparation of Positive Electrode>

A band-like positive electrode having an electrode density of 3.0 g/cm³ was prepared by dissolving 91% by weight of a lithium-containing composite oxide powder having a composition equal to that described previously in conjunction with Example 1, 2.5% by weight of an acetylene black, 3% by weight of graphite, and 4% by weight of polyvinylidene fluoride (PVdF) in N-methyl pyrrolidone (NMP), followed by coating an aluminum foil having a thickness of 15 μ m with the resultant solution and subsequently drying and, then, pressing the coating.

<Preparation of Negative Electrode>

A band-like negative electrode having an electrode density of 1.4 g/cm³ was prepared by dissolving 94% by weight of mesophase pitch based carbon fibers having an average fiber diameter of 25 μ m and an average fiber length of 30 μ m, which had been subjected to a heat treatment at 3,000°C and 6% by weight of polyvinylidene fluoride (PVdF) in N-methyl pyrrolidone (NMP), followed by coating a copper foil having a thickness of 12 μ m with the resultant solution and

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subsequently drying the coating and, then, pressing the coating.

<Preparation of Electrode Group>

A laminate structure comprising the positive electrode noted above, a separator formed of a polyethylene porous film having a thickness of 16 μ m, a porosity of 50%, and an air permeability of 200 seconds/100 cm³, the negative electrode noted above, and the separator noted above, which were laminated in the order mentioned, was spirally wound. The wound structure was thermally pressed at $90^{\circ}\mathrm{C}$ so as to prepare a flat electrode group having a width of 30 mm and a thickness of 3.0 mm. The electrode group thus prepared was housed in a laminate film bag formed of a laminate film having a thickness of 0.1 mm and comprising an aluminum foil having a thickness of 40 $\mu\,\mathrm{m}$ and a polypropylene layer formed on each of both surfaces of the aluminum foil. The electrode group housed in the laminate film bag was subjected to vacuum drying at 80° C for 24 hours.

<Preparation of Nonaqueous Electrolyte (Liquid
Nonaqueous Electrolyte)>

A nonaqueous electrolyte was prepared by dissolving lithium tetrafluoroborate (LiBF4) used as a solute in a mixed solvent prepared by mixing ethylene carbonate (EC), γ -butyrolactone (BL) and vinylene carbonate (VC) in a volume ratio of 24 : 75 : 1. The

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solute was dissolved in the mixed solvent in an amount of 1.5 mol/L.

After the nonaqueous electrolyte noted above was poured into the laminate film bag having the electrode group housed therein, the laminate film bag was completely sealed by a heat seal so as to prepare a thin lithium ion secondary battery constructed as shown in FIGS. 2 and 3 and having a width of 35 mm, a thickness of 3.2 mm and a height of 65 mm.

Example 122:

A thin lithium ion secondary battery was prepared as in Example 121, except that used as the positive electrode active material was a lithium-containing composite oxide having a composition equal to that described previously in conjunction with Example 27.

Example 123:

A thin lithium ion secondary battery was prepared as in Example 121, except that used as the positive electrode active material was a lithium-containing composite oxide having a composition equal to that described previously in conjunction with Example 50.

Example 124:

A thin lithium ion secondary battery was prepared as in Example 121, except that used as the positive electrode active material was a lithium-containing composite oxide having a composition equal to that described previously in conjunction with Example 70.

Example 125:

A thin lithium ion secondary battery was prepared as in Example 121, except that used as the positive electrode active material was a lithium-containing composite oxide having a composition equal to that described previously in conjunction with Example 90.

Example 126:

A thin lithium ion secondary battery was prepared as in Example 121, except that used as the positive electrode active material was a lithium-containing composite oxide having a composition equal to that described previously in conjunction with Example 106.

<Nail Sticking Test>

A nail sticking test was applied to the secondary battery prepared in each of Examples 121 to 126. In the first step, each of these secondary batteries was charged. Specifically, the battery was charged to 4.2 V under a current value corresponding to 0.2C based the rated design capacity of the secondary battery and, then, maintained constant at 4.2 V. The charging was performed for 8 hours in total. After the charging to 4.2 V, the safety of the secondary battery was studied by applying a nail sticking test. The nail used in the test had a diameter of 2 mm, and the nail speed was set at 135 mm/sec. Also, the temperature elevation of the secondary battery in the nail sticking test was measured by a thermocouple attached on the outer

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surface of the secondary battery. Table 14 shows the occurrence or nonoccurrence of rupture and ignition in the nail sticking test and the battery temperature in the nail sticking test.

<Discharge Capacity Reduction Rate after 300
Cycles>

A charge-discharge cycle test was applied at room temperature to the secondary battery prepared in each of Examples 121 to 126 so as to obtain the reduction rate of the discharge capacity after 300 cycles.

Table 14 shows the results. In the charge-discharge cycle test, the battery was charged to 4.2 V under a current corresponding to 0.5C based the rated design capacity, and the constant voltage of 4.2 V was maintained. The charging was performed for 5 hours in total. On the other hand, the battery was discharged to 2.7 V at the same current. Also, a rest time of 30 minutes was provided between the charging and the discharging.

<Large Current Discharge Characteristics
(Discharge Rate Characteristics)>

The secondary battery for each of Examples 121 to 126 was charged to 4.2 V under a current corresponding to 0.5C based the rated design capacity and, then, the voltage was maintained constant at 4.2 V. The charging was performed for 5 hours in total. Further, the battery was discharged 30 minutes later to 2.7 V at

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a current corresponding to 0.5C so as to measure the initial discharge capacity, which is equal to the discharge capacity in this discharge operation. Then, the secondary battery was charged to 4.2 V at a current corresponding to 0.5C based the rated design capacity, followed by maintaining the constant voltage of 4.2 V. The charging was performed for 5 hours in total. secondary battery was discharged again 30 minutes later to 2.7 V at a current corresponding to 5C. discharge capacity at the time when the secondary battery was discharged at 5C was compared with the initial discharge capacity so as to obtain the rate (%) of the reduced capacity at the discharge at 5C relative to the initial discharge capacity. Table 14 also shows the result as the high rate discharge characteristics (discharge rate characteristics).

<High Rate Cycle Characteristics under the
Condition of Large Current Discharge>

The secondary battery for each of Examples 121 to 126 was charged to 4.2 V at a current corresponding to 0.5C based the rated design capacity and, then, the voltage was maintained constant at 4.2 V. The charging was performed for 5 hours in total. Further, the secondary battery was discharged 30 minutes later to 2.7 V at a current corresponding to 5C. The discharge capacity was measured after the charge-discharge operations noted above were repeated 100 times so as to

obtain the rate of the reduced capacity at the discharge time after 100 cycles relative to the initial discharge capacity. Table 14 also shows the results as the high rate cycle characteristics.

		Table 14			
		hatterv	Reduction rate	High rate	High rate
	Result of hall scieking	- 1170	of discharde	discharge	cycle
	test	in nail	Ü.	characteri	character-
		sticking	300 cycles	stics	istics
		test	(%)	(%)	(%)
		(2)			
			α 1	-14	-20
Example 121	No rupture and no ignition	QOT	Ì	7	-21
1	The same of the first tion	80	J)	174	1
Example 122	No rupture	68	8	-13	-21
Example 123	No rupture	70	ì	1.3	-21
Example 124	No rupture and no ignition	78	0 1	2	C
		75	-111	J)	5
Example 125	No rupture		0.1	6	ი 1
Fxample 126	No rupture and no ignition	69	77.		
o talling of					

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As apparent from Table 14, the thin secondary batteries for Examples 121 to 126 containing as active materials the lithium-containing composite oxides having compositions represented by chemical formulas (1) to (6) referred to previously were free from rupture and ignition in the nail sticking test, capable of suppressing temperature elevation and excellent in safety. Also, the thin secondary batteries for Examples 121 to 126 were found to be excellent in charge-discharge cycle characteristics, high rate discharge characteristics, and high rate cycle characteristics.

Each of the lithium ion secondary batteries for the Examples noted above contains an electrolyte prepared by using an organic solvent, making it necessary to suppress the rupture and ignition.

In order to suppress rupture and ignition, it is desirable to set the temperature elevation of the battery at 150°C or lower, more desirably, at 110°C or lower, though the temperature elevation in question is dependent on the kind of the organic solvent used for preparing the electrolyte. If the temperature elevation exceeds 110°C, it is possible for rupture and ignition to be caused. In Examples 1 to 126, the battery temperature was not higher than 110°C and, thus, rupture and ignition of the secondary battery were not brought about. On the other hand, the battery

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temperature was rapidly elevated to 400°C or more in Comparative Examples 2, 4 and 6 to 13, leading to the rupture and ignition of the secondary battery.

Incidentally, the Examples described above are directed to a cylindrical lithium ion secondary battery and a thin plate type lithium ion secondary battery. However, the lithium ion secondary battery of the present invention is not limited to a cylindrical battery and a thin plate type battery. It is also possible to apply the technical idea of the present invention to, for example, a rectangular or button type lithium ion secondary battery in addition to the cylindrical lithium ion secondary battery and the thin plate type lithium ion secondary battery. It is also possible to use a laminate film as a casing in place of a metal can.

As described above in detail, the present invention provides a positive electrode active material capable of preventing rupture and ignition in the nail sticking test and also capable of improving the high rate discharge characteristics (discharge rate characteristics), and a lithium ion secondary battery utilizing the particular positive electrode active material.

Additional advantages and modifications will readily occur to those skilled in the art. Therefore, the present invention in its broader aspects is not

limited to the specific details and representative embodiments shown and described herein. Accordingly, various modifications may be made without departing from the spirit or scope of the general inventive concept as defined by the appended claims and their equivalents.